Reversible Topochemical Polymerization and Depolymerization of a Crystalline Three-Dimensional Porous Organic Polymer with C–C Bond Linkages

Chenyue Sun  
MIT  https://orcid.org/0000-0002-7524-5323

Julius Oppenheim  
MIT

Grigori Skorupskii  
Massachusetts Institute of Technology  https://orcid.org/0000-0002-7089-2724

Luming Yang  
MIT

Mircea Dinca (mdinca@mit.edu)  
Massachusetts Institute of Technology  https://orcid.org/0000-0002-1262-1264

Article

Keywords: polyMTBA, crystallinity, topochemical polymerization

Posted Date: October 28th, 2021

DOI: https://doi.org/10.21203/rs.3.rs-900095/v1

License: This work is licensed under a Creative Commons Attribution 4.0 International License. Read Full License
Reversible Topochemical Polymerization and Depolymerization of a Crystalline Three-Dimensional Porous Organic Polymer with C–C Bond Linkages

Chenyue Sun, Julius J. Oppenheim, Grigori Skorupskii, Luming Yang, Mircea Dincă*
Department of Chemistry, Massachusetts Institute of Technology, Cambridge, Massachusetts 02139, United States
Email: mdinca@mit.edu

Abstract
Three-dimensionally connected porous organic polymers are of interest because of their potential in adsorption, separation, and sensing, among others. When crystalline, they also afford accurate structure description, which in turn can enable particular functions. However, crystallization of three-dimensional (3D) polymers is challenging. This is especially true when targeting polymerization via stable C–C bonds, whose formation is usually irreversible and does not allow for error correction typically required for crystallization. Here, we report polyMTBA, the first 3D-connected crystalline organic polymer with permanent porosity, here formed via C–C linkages. High crystallinity is achieved by solid-state topochemical reaction within monomer MTBA crystals. polyMTBA is recyclable via thermal depolymerization and is solution-processable via its soluble monomers. These results reveal topochemical polymerization as a compelling methodology for generating stable, crystalline, and porous 3D organic frameworks.

Main
Organic porous materials such as hypercrosslinked polymers,1 covalent organic frameworks (COFs),2 conjugated microporous polymers (CMPs),3 and porous aromatic frameworks (PAFs)4 have garnered considerable attention for their utility in adsorption,5–9 catalysis,10–12 and energy storage.13–18 Among these, COFs stand out because of their crystallinity, which both allows for accurate description and understanding of their structure, and confers unique benefits such as monodisperse size-selective sieving, shape selection, and control over capillary condensation.19–21 On the other hand, CMPs, PAFs, and hypercrosslinked polymers stand out through their typically enhanced chemical stability afforded by irreversible C–C and aromatic linkages.22–25 Despite recent advances in C=C-linked two-dimensional (2D) COFs,26 it is still challenging to form crystalline C–C-linked porous frameworks via solvothermal routes because these require irreversible bond formation and crystallization in a single step, which is incompatible with error correction during crystal growth.

To this end, topochemical polymerization provides a promising avenue to stable, highly crystalline materials.27 With topochemical polymerization, monomer polymorphs are chosen in such a way that the reactive functional groups are spatially aligned in close proximity to allow for bond formation in the solid state. Subsequent C–C bond formation, triggered by light or some other stimuli, is effectively decoupled from crystallization and thus gives rise to highly crystalline polymers. Although early examples of topochemical polymerization primarily focused on generating one-dimensional polymers, particularly with diynes and dienes,27,28 in recent years, increasing efforts have been devoted to 2D polymers, taking advantage of the photochemical dimerization reaction of stilbene and anthracene derivatives.29–33 To our knowledge, however, this methodology has not yet been extended to produce 3D crystalline organic polymers.
Here, we report that photochemical polymerization of methanetetrayltetrakis(benzene-4,1-diyl) tetrakis(anthracene-9-carboxylate) (MTBA, Fig 1a) crystals in the solid state yields a crystalline porous solid (polyMTBA, Fig. 1f) with permanent porosity. Even though MTBA crystals have no discernible crystallographic voids, porosity emerges in polyMTBA due to a large displacement of monomer molecules during polymerization, which creates one-dimensional channels along the c-direction (Fig 1e-f). polyMTBA can be deposited onto glass substrates from solution-processed MTBA films, and can be thermally degraded to recover pristine monomer.

The molecular structure of MTBA was carefully chosen to favor a rigid and open framework structure. Anthracene is attractive as a topochemical motif because of its relatively small size and its well-documented photodimerization chemistry. In particular, 9-substituted anthracenes tend to crystallize and dimerize in head-to-tail fashion (Fig. 1b). Inspiration for the tetrahedral tetraphenylmethane core in MTBA came from COFs, which use this motif to impose high rigidity and maximal angular separation between the peripheral groups, which are conducive to permanent porosity. It should be noted that due to the highly congested nature around the bridgehead carbon atoms in dianthracene, the range of permissible substituents at the 9 and 10 position of anthracene is quite limited. For instance, 9,10-diphenylanthracene is photochemically inert in both solution and solid state, likely because the ortho-H on the phenyl group prevents the proximity of two anthracene units in the transition state. A sterically less demanding ester linkage, on the other hand, does not affect the dimerization.

MTBA crystallizes in the tetragonal space group I41/a with no lattice solvent. Individual MTBA molecules stack into columns along the c-direction, collinear with the S₄ molecular axis (Fig. 1d). Each anthracene group is paired face-to-face (dihedral angle < 0.01°, interplanar distance 3.3 – 3.6 Å) with an anthracene from a neighboring molecule. However, significant slipping displacement between the two neighboring anthracene groups gives rise to a large C₉₋₋C₁₀’ separation of 5.3-5.6 Å (Supplementary Fig. 1), which is much larger than what is commonly believed to be the maximal separation (4.2 Å, Schmidt criterion) allowed for topochemical anthracene dimerization. However, the two neighboring anthracene groups still interact electronically and give rise to excimeric emission, as exhibited by the green photoluminescence (λmax = 503 nm, Supplementary Fig. 2) of a polycrystalline powder of MBTA.
Despite the large C9–C10’ separation, irradiation of MTBA crystals at 370 nm or 427 nm induced a noticeable change of the fluorescence from intense green to faint sky-blue, while the solid turned from bright yellow to beige. In our hands, the most complete transformation, as assessed by adsorption capacity (vide infra), was obtained by irradiating a continuously stirred suspension of powder MTBA in diethyl ether under 1 bar of CO₂ with a 427 nm LED for 3 days. The absence of CO₂ during irradiation leads to a product exhibiting lower porosity, leading us to believe that CO₂ may fill pores generated during topochemical transformation, thereby compensating for the loss of dispersive interaction between MTBA molecules. Additionally, MTBA itself adsorbs 13.3 cm³/g CO₂ at 1 bar and 273 K, confirming that CO₂ penetrates the monomer lattice to some extent (Supplementary Fig. 3). Importantly, diethyl ether does not dissolve MTBA and therefore serves as a good suspension medium for monomer powder, allowing for better irradiation exposure. Furthermore, 427 nm is at
the edge of the absorption peak of MTBA (Supplementary Fig. 4); irradiation at this wavelength improves light penetration into the crystals.\textsuperscript{40}

![Image of graphs showing intensity vs. angle and chemical shift vs. ppm](image)

**Fig. 2.** Qualitative and quantitative evidence for transformation from MTBA to polyMTBA. **a**, Transformation is evident from significant changes in the intensity and peak positions of the powder X-ray diffraction (PXRD) patterns. **b**, Quantitative $^1$H solid-state nuclear magnetic resonance (SS-NMR) spectra show clear appearance of the bridgehead benzylic proton (denoted by an arrow) upfield of the aromatic signals. Strong intermolecular shielding causes large upfield shifts. Asterisks denote artifact signals from sample rotor/probe.

Characterization of the photoproduct suggests the emergence of a significant amount of dianthracene linkages and loss of spectroscopic features for monomeric anthracene groups. Diffuse-reflectance UV-visible spectroscopy shows significant weakening of the characteristic absorption of anthracene above 300 nm in the photoproduct (Supplementary Fig. 4). By diffuse-reflectance infrared Fourier-transform spectroscopy (Supplementary Fig. 5), MTBA shows no strong features in the 700 – 600 cm\textsuperscript{-1} region, whereas the photoproduct displays two strong bands at 690 cm\textsuperscript{-1} and 636 cm\textsuperscript{-1}, characteristic of anthroate dimers.\textsuperscript{41} A cross-polarized magic-angle spinning $^{13}$C SS-NMR spectrum of the photoproduct displays new peaks at approximately 50 ppm, which are assigned to the bridgehead carbon in dianthracene (Supplementary Fig. 6). Altogether, vibrational, electronic, and NMR spectroscopy along with the loss of excimeric emission confirm the dimerization of a significant portion of neighboring anthracene units in solid MTBA.

To quantify the degree of photopolymerization, we turned to high-resolution solid-state $^1$H-NMR spectroscopy with homonuclear dipolar decoupling.\textsuperscript{42} We note that in the solid state, all protons shift significantly upfield relative to the solution spectra due to the strong shielding effect of nearby aromatic rings,\textsuperscript{43} but the benzylic proton of dianthracenes can still be readily discerned from the aromatic protons (Fig. 2b). Peak integration gives a ratio of 12.8:1 aromatic to benzylic protons, which equates to approximately 93\% polymerization. This high degree of photodimerization is surprising given that the separation of anthracene units in MTBA seems to violate the Schmidt criterion (vide supra), but is by no means unprecedented and is likely enabled by significant motion of the monomers under irradiation: solid-state photodimerization reactions involving even larger molecular motions have been reported.\textsuperscript{44,45} Indeed, the relatively slow photopolymerization in our system (days, cf. hours in ref.\textsuperscript{30,31}) is in line with those involving significant molecular displacement. If we assume that all anthracene groups within each monomer engage in photodimerization, the photoproduct is a 3D polymer network, i.e. polyMTBA (Figs. 1c and 1d). Consistent with this assignment, the photoproduct is insoluble in common organic solvents.
Crucially, the transformation from **MTBA** to **polyMTBA** preserves crystallinity. The powder X-ray diffraction (PXRD) pattern of **polyMTBA** is clearly distinct from that of **MTBA** in both peak positions and relative intensities (Fig. 2a, Supplementary Fig. 8). Scanning electron microscopy (SEM) showed that the square prismatic shape of MTBA crystallites persists after polymerization without noticeable deformation (Supplementary Fig. 7). However, efforts to obtain single crystals of **polyMTBA** were unsuccessful: even when attempting photopolymerization at low temperature (223 K), single crystals of **MTBA** rapidly degrade to show only polycrystalline diffraction features. However, a combination of Pawley refinement and molecular mechanics optimization provided reliable insight into the structure of **polyMTBA** (Fig. 1, Supplementary Figs. 8 and 9). Thus, high-resolution synchrotron PXRD data of **polyMTBA** can be fitted well in the same $I\bar{4}_1/a$ space group as the monomer, with notable expansion along both $a$ (28.8432(3) Å vs 28.2999(3) Å of **MTBA**) and $c$ (15.9068(2) Å vs 15.0898(2) Å of **MTBA**). Scaling the experimental crystal structure of **MBTA** accordingly and replacing the anthracene monomer pairs with dianthracenes gave a polymer model used as the input for molecular mechanics optimization. Overall, the computed fully polymerized **polyMTBA** can be viewed as a six-fold interpenetrated diamond-type framework (Fig. 1c). Importantly, one-dimensional open channels with a diameter of approximately 3.9 Å extend parallel to the $c$-direction (Fig. 1f, Supplementary Fig. 9).

Fig. 3. Characterization of surface-deposited **MTBA** and **polyMTBA**. a, Crystallites of **polyMTBA** on a glass substrate under optical microscope. b, Grazing incidence X-ray diffraction pattern before and after irradiation, showing the (200) diffraction shifting to lower angle. c, UV-visible absorption spectra of **MTBA** deposited on quartz before and after irradiation. d, Photoluminescence spectra of the same sample as in c.

Thanks to the solubility of **MTBA**, **polyMTBA** can be facilely processed into various forms. For instance, spin-coating of an **MTBA** solution onto glass and quartz substrates forms initially amorphous films, which transform into discrete spindle-shaped crystallites upon solvent-vapor annealing. Subsequent irradiation of this surface-deposited **MTBA** precursor led to evident shifts in the grazing-incidence X-ray diffraction pattern, indicative of polymerization and formation of **polyMTBA** (Fig. 3b) without apparent change in the crystallites (Fig. 3a). Surface deposition here allows for efficient light penetration and a high degree of polymerization after
irradiation (Fig. 3a, Supplementary Fig. 10) as evidenced by the clean disappearance of the anthracene UV-visible absorption above 300 nm (Fig. 3c) and complete loss of excimeric photoluminescence (Fig. 3d).

CO\textsubscript{2} adsorption isotherms confirmed the ultramicroporous nature of polyMTBA. Due to kinetic restrictions, polyMTBA adsorbs minimal N\textsubscript{2} at 77 K.\textsuperscript{46} However, a CO\textsubscript{2} isotherm measured at 195 K exhibits steep uptake at low pressure and a capacity of 108 cm\textsuperscript{3}/g STP (4.75 mmol/g, 14.8 wt\%) at 0.93 bar (Supplementary Fig. 11). At 273 K, polyMTBA adsorbs 31.3 cm\textsuperscript{3}/g STP (1.38 mmol/g, 4.8 wt\%) at 1.0 bar (Supplementary Figs. 11 and 12). This capacity is very close to the value obtained from simulation on the molecular mechanics-optimized model\textsuperscript{47} (34.2 ± 2.0 cm\textsuperscript{3}/g STP at 1.0 bar). Fitting the CO\textsubscript{2} isotherm at 195 K to the BET model gives an apparent surface area of 350 m\textsuperscript{2}/g (calc. 408.6 ± 0.4 m\textsuperscript{2}/g for the optimized model).

Importantly, polyMTBA depolymerizes into MTBA at elevated temperature. Thermogravimetric analysis (TGA) of polyMTBA and MTBA under inert atmosphere gives identical results, with a plateau at lower temperature and a rapid mass loss above 375 °C (Supplementary Fig. 13). The similar thermal behavior between the monomer and the polymer is consistent with the expectation that polyMTBA depolymerizes cleanly below 375 °C. To pinpoint the decomposition temperature, we performed variable-temperature (VT) PXRD of polyMTBA under a helium atmosphere. The characteristic diffraction peaks of the polymer remain largely unchanged below 90 °C, above which diffraction peaks start losing intensity (Supplementary Fig. 14a). At much higher temperature, the residual weak diffraction strongly resembles that of MTBA (Supplementary Fig. 14b). Indeed, an NMR spectrum of the thermolyzed sample is identical to that of pristine MTBA (Supplementary Figs. 15-16). Consistent with the VT-PXRD data, differential scanning calorimetry (DSC) of polyMTBA also reveals exothermic transformations with an onset temperature of approximately 100 °C (Supplementary Fig. 17), albeit displaying complexities that are in line with the behavior of previously reported 2D topochemical polymers.\textsuperscript{30} Altogether, the TGA, VT-PXRD, and DSC data pin 100 °C as the onset of thermal depolymerization. Under ambient conditions, minimal depolymerization is observed over weeks (Supplementary Fig. 18).

Conclusions

The foregoing results describe polyMTBA, a rare example of a 3D-connected, porous crystalline organic polymer. Enabled by topochemical photopolymerization, the formation of polyMTBA as a crystalline product is possible because crystallization of the monomers and the irreversible C–C bond formation between the monomers are decoupled. We show that topochemical dimerization of anthracene is possible despite a large C\textsubscript{9}–C\textsubscript{10}' separation of 5.3-5.6 Å, and that one-dimensional channels absent in the monomer crystals are created upon polymerization. The synthetic strategy introduced here enables the material to be recyclable and solution-processable, thereby potentially serving as a blueprint for generating other complex and porous frameworks, including by exploiting co-crystallization and co-polymerization of dissimilar monomers.

Data availability

Experimental and computational details and additional spectroscopic and crystallographic characterization data supporting findings in this paper are available in the Supplementary Information. Crystallographic information for MTBA has been deposited in the Cambridge Crystallographic Data Centre under accession code CCDC 2084357. Computed structure of polyMTBA is also provided as a CIF file.
References


**Acknowledgements**

The high-resolution PXRD data was collected at 11-BM of the Advanced Photon Source at Argonne National Laboratory, which was supported by the U. S. Department of Energy, Office of Science, Office of Basic Energy Sciences, under Contract No. DE-AC02-06CH11357. We are thankful to Prof. Marc Baldo and Dr. Dong-Gwang Ha for help with photoluminescence measurements.

**Author Contribution**

C.S. and M.D. designed the experiments. C.S. executed the synthesis and characterization. C.S. and J.O. analyzed the adsorption data. G.S. performed crystallographic data collection and analysis. L.Y. collected and analyzed mass spectrometry data as well as analyzed nuclear magnetic resonance data. All authors contributed to writing of the manuscript and have given approval to the final version of the manuscript.

**Competing interests**

The authors declare no competing interests.
Supplementary Files

This is a list of supplementary files associated with this preprint. Click to download.

- 2084357.cif
- polyMTBAcomputedstructure.cif
- SupportingInformation.pdf