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**Efficient photodegradation of Azucryl Red by copper doped TiO₂
nanoparticles – Experimental and modelling studies**

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Abstract

This work aims to investigate the effect of copper doping on the photocatalysis performance of TiO₂ nanoparticles for disposal wastewater from organic pollutants. X-ray diffraction analysis manifests the crystallization of a rutile phase for pure and copper-doped TiO₂ except for 2% resulting in a rutile-to-anatase phase transformation. The crystallite size is found less affected by Cu doping; i.e. ~30 nm. BET analysis indicates a decrease in the specific surface area as the doping loading increases. Scanning electron microscopy observations reveal spherical particles at the nanometer range for pure TiO₂ then larger agglomerates of ultrafine particles with Cu doping. FTIR analysis notifies the existence of hydroxyl groups that will promote the photocatalysis process. The photodegradation of Azucryl Red (AR) has been investigated under different conditions; i.e., Cu-loading, initial concentration of AR, and pH. The kinetics of the photodegradation process is further found to comply with the Lagergren kinetic law, regardless the experimental conditions. Nevertheless, the photodegradation process is not only controlled by the intra-particle diffusion mechanism, but also by mass transfer through a liquid film boundary. The maximum degradation of AR, i.e. 86%, has been achieved at pH = 5.0 during 60 minutes of contact time for the 2% Cu-doping, with effective regeneration. The Freundlich model exhibits a better fitting for AR dye photodegradation equilibrium data, compared to Langmuir, Temkin and Dubinin-Radushkevich.

Keywords: Sol-gel; Cu-doping; Rutile; Anatase; Phase transformation; Photodegradation; Azucryl red.

1. Introduction

Industrial activities generate significant volumes of water contaminated by polluting organic and inorganic matter. The organic contaminants include synthetic dyes in the industrial sector that are widely used in the paper industry, cosmetics, food and especially in the textile industry (El Haddad, Regti, Laamari, Mamouni, & Saffaj, 2014; Mahdavinia & Zhalebaghy, 2012; Perez-Urquiza, Prat, & Beltran, 2000). These non-biodegradable dyes are likely to contaminate the environment, besides being responsible for various environmental alterations, especially water resources, their presence in the water, their pollution rate and their problem of water pollution.

Although several techniques have been adopted for organic dyes remediation, the heterogeneous photocatalysis currently appears as a very promising and potential technological alternative. Indeed, during the process of degradation, there will be the destruction of organic pollutants, such as dyes, while leading to a complete mineralization (Fernández-García et al., 2000; Joshi & Shrivastava, 2011). It has been reported that the photocatalytic process depends mainly on the characteristics of the molecule to be degraded, the light source and the type of semiconductor acting as a photocatalyst (gap and texture). In the literature, the widely investigated photocatalysts are zinc oxide (ZnO) and titanium dioxide (TiO₂) (Hoffmann, Martin, Choi, & Bahnemann, 1995; Kormann, Bahnemann, & Hoffmann, 1988; Liu & Yang, 2003; Pelizzetti, Minero, Piccinini, & Vincenti, 1993; Villaseñor, Reyes, & Pecchi, 1998; William IV, Kostedt, Ismail, & Mazyck, 2008).

Titanium dioxide is currently considered to be the most efficient semiconductor owing to its enhanced photocatalytic performance, non-toxicity, low cost, durability and chemical stability to corrosion (Hoffmann et al., 1995), as well as for its large and tunable bandgap (3.2 eV for anatase and 3.0 eV for rutile). Moreover, it uses UV rays containing only a small part (about 3 to

5%) of solar energy (Linsebigler, Lu, & Yates Jr, 1995). Many efforts have recently been devoted to optimize the characteristics of TiO₂ in the visible (Wu, Dong, Zhao, & Guo, 2008) such as through the doping TiO₂ with transition metal ions (Anpo & Takeuchi, 2003; Chen et al., 2002; Einaga, Harada, Futamura, & Ibusuki, 2003; Thaminimulla, Takata, Hara, Kondo, & Domen, 2000; Xie & Yuan, 2003) and non-metallic anions (Asahi, Morikawa, Ohwaki, Aoki, & Taga, 2001; Burda et al., 2003; Gómez et al., 2003; Yu, Yu, Ho, Jiang, & Zhang, 2002). Moreover, other approaches have been also proposed, such as the coupling with narrow-band semiconductors (Bartl, Boettcher, Frindell, & Stucky, 2005; Bartl, Puls, Tang, Lichtenegger, & Stucky, 2004; Kumar & Jain, 2001) and sensitization with dyes (Viseu, Hungerford, & Ferreira, 2002). Besides, it is well-known that the doping process of TiO₂ by metals and transition metals is much easier than using non-metallic elements (Iwaszuk & Nolan, 2011).

This research work, aims to investigate the influence of Cu as an effective transition metal dopant in order to improve TiO₂ photocatalytic properties (Srinivas et al., 2011). TiO₂ and Cu-doped nanopowders are prepared using the sol-gel method then characterized by XRD, SEM/EDX, BET, and FTIR. The photocatalytic performance of the as-prepared nanopowders is studied using azucryl red (AR) molecule as a textile dye. The photodegradation data were modeled using different kinetics and equilibrium models. A mechanism of the photodegradation process was proposed and discussed.

2. Experimental Part

2.1. Synthesis of Photocatalysts

Pure and Cu-doped TiO₂ powders are prepared via a cost-effective and straightforward procedure. The precursor tetraethyl-orthotitanate ((C₂H₅O)₄Ti, Merk 95%) was dissolved in ethanol, water, and nitric acid (HNO₃ Biochem 69%) solution. The mixture was roughly stirred

using a magnetic agitator for one hour. After 24 hours of soaking, the mixture was oven-dried at 100°C and continuously stirred in air. The obtained powder was amorphous. The sol of Cu-doped TiO₂ was prepared using the identical method as previously detailed, with the addition of copper nitrate (Cu(NO₃)₂) to the TiO₂ solution. The as-produced powders are subsequently calcinated for one hour at 600°C.

2.2. Characterization of Photocatalysts

The structure of the as-prepared powders was checked by X-ray powder diffraction (XRD) using Rigaku Ultima IV diffractometer with Cu-K α X-ray source ($\lambda = 1.5418 \text{ \AA}$). The morphological examinations have been accomplished by field emission scanning electron microscopy (FESEM) employing LYRA3 TESCAN provided with a dispersive electron X-ray (EDX) spectrometer for the specification of the chemical composition. The BET (The Brunauer, Emmett and Teller) specific surface area (SSA), average pore diameter and volume were determined from nitrogen adsorption-desorption isotherms performed at the boiling point (-195.8°C) of N₂ through an ASAP 2020 (Micromeritics) apparatus. Each sample was outgassed using He for 2h at 250°C to take off moistness and impurities prior to the adsorption experiment. The BET model and t-plot scheme of Lippens and de Boer (Lippens & De Boer, 1965) were used to calculate the porosity and surface area. The FTIR chemical bonding were analyzed using an Nicolet 6700 spectrometer, FTIR spectra were achieved in the interval of 4000–400 cm⁻¹ with a resolution of 4 cm⁻¹.

2.3. Photocatalytic Degradation Measurements

The photocatalysis test was carried out at atmospheric pressure and room temperature in a catalytic photoreactor, as described elsewhere (Kerrami et al., 2019). 0.2 mg of catalyst was added

to 2 mL of azucryl red (AR) solution having an initial concentration $C_0 = 1,247 \times 10^{-5} \text{ mol.L}^{-1}$ in a quartz container. The heterogeneous mixture was then irradiated with UV light with a cut filter (365 nm) using a UV light lamp (Spot Light Source 300-450nm, L9588-01, Hamamatsu, Japan) and kept under magnetic agitation. The absorbance was measured for different exposure periods of time using a spectrophotometer scan (39919.FEB /10) at a maximum absorption wavelength of about 532 nm.

The amount of adsorbed AR dye was calculated according to the formula:

$$q_t = \frac{V(C_0 - C_t)}{m} \quad (1)$$

where q_t (mg.g^{-1}) is the adsorbed dye amount per gram of adsorbent at time t (min), V is the solution volume (L), C_0 and C_t are the initial concentration and concentration of AR dye (mg.L^{-1}) in solution at any time t , and m the mass of the adsorbent (g).

The quantity adsorbed (mg dye/g material) at equilibrium q_e was estimated through an analogue expression:

$$q_e = \frac{V(C_0 - C_e)}{m} \quad (2)$$

where C_e is the liquid-phase concentrations of AR dye at equilibrium time.

3. Results and Discussion

3.1 X-ray diffraction analysis

Figure 1 illustrated the evolution of X-ray diffraction patterns of pure and Cu 2-10% doped TiO_2 . Qualitative phase analysis indicates the formation of rutile phase for pure TiO_2 , followed by phase transformation into anatase phase for $\text{Cu}_{2\%}$, which in turn undergoes a second phase

transformation into original rutile phase for Cu_{4%} and Cu_{10%} doped TiO₂. Figure 1 also displays that all diffraction peaks are comparatively broad and intense, implying the formation a nanoscale phase with high crystallinity. For pure TiO₂, the observed peaks at $2\theta = 27, 36, 41, 44, 54, 56, 62, 64$ and 69° are indexed as (110), (101), (111), (211), (220), (002), (311), (301), and (112) reflections of the tetragonal rutile phase, in conformity with JCPDS card No. JCPDS card No. 01-075-1750. For Cu_{2%} doped TiO₂, the newly emerged diffraction peaks detected at $2\theta = 25.3, 37.8, 48.1, 53.9, 55.1, 62.7, 68.8, 70.4$ and 75.07° are indexed as (101), (004), (200), (105), (211), (204), (116), (220), and (215) reflections of the tetragonal TiO₂ (anatase phase) structure in accordance with JCPDS card No. 00-021-1272 (Pongwan, Wetchakun, Phanichphant, & Wetchakun, 2016). Whereas for Cu_{4%} and Cu_{10%} doping contents, all the diffractions peaks belong to the tetragonal rutile phase similarly to pure TiO₂. However, it is important to mention that for Cu doping in the range 4–10%, new low intensity diffraction peak appears around of $2\theta = 44^\circ$ attributed to the (111) reflection belonging to Cu metal in agreement with JCPDS card No. 02-1225 (Del Carro, 2018).

In a similar work, Park et al. reported that anatase-rutile phase transformation of TiO₂ occurred at 2.5 wt.% Cu doping using low temperature process at 60 °C for 4 and 14 h (Park, Kim, Kim, & Lee, 2006). Besides, due to the differences between the ionic rays (Ti⁴⁺ = 0.64 Å and Cu²⁺ = 0.73 Å) and the cationic charges of Ti (+4) and Cu (+2), oxygen vacancies may be formed for maintaining charge neutrality (Sahu & Biswas, 2011).

The average crystallite size (D) of Cu_{x%}-doped TiO₂ nanoparticles have been calculated by using the most intense peak (101) anatase and (110) rutile by applying the Scherrer equation:

$$D = \frac{K \lambda}{\beta \cos \theta} \quad (3)$$

where λ is the X-ray wavelength (1.5417 Å), θ is the Bragg angle, and β is the full width at half maximum of the corresponding peak.

According to Williamson–Hall approach (Zak, Majid, Abrishami, & Yousefi, 2011), strain and crystallite size contribute significantly to the diffraction lines broadening (equation 4) (Bindu & Thomas, 2014; Pandiyarajan & Karthikeyan, 2012). In the uniform deformation model (*UDM*), it is assumed that a crystal is isotropic and subsequently its properties are independent of the crystallographic direction along which the measurement is considered.

$$\beta_{hkl} \cos \theta_{hkl} = \frac{k\lambda}{D} + 4\varepsilon \sin \theta_{hkl} \quad (4)$$

A plot of $\beta_{hkl} \cos \theta_{hkl}$ versus $4 \sin \theta_{hkl}$ represents a linear graph (Ungár, 2007), whereas microstrain (ε) and the crystallite size (D) are calculated from the slope and intercept of the linear plot, respectively.

The d-spacing and the lattice parameters have been estimated by following expressions:

$$d = \frac{\lambda}{2 \sin \theta} \quad (5)$$

$$\frac{1}{d^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2} \quad (6)$$

All calculated structural (d-spacing and lattice parameters) and microstructural (crystallite size and microstrain) parameters are given in Table 1. The mean crystallite size values estimated using the Scherrer equation and Williamson–Hall model are virtually identical, signifying that the inclusion of strain in the modified formulae of W–H method has inconsequential impact on the average (D) value. Yet, the average crystallite size estimated using Scherrer’s and W–H methods exhibit little disparity that can be ascribed to the variation in particle size distribution averaging (Zak et al., 2011). Both values of crystallite size have the same trend, with 4% Cu doping exhibiting the smallest crystallite size of 26 (29) nm and the lowest microstrain with $48 \times 10^{-5} \%$.

The lattice parameters of pure TiO_2 are found to be vary anisotropically compared to the standard values of rutile phase; a and b increase from 4.5912 to 4.5937 while c decreases from

2.9599 to 2.9587 Å without any significant variation in the unit cell volume; i.e. 62.39 compared to 62.43 Å³. Concerning the influence of Cu doping on the lattice parameter, it can be noticed that for 2% Cu loading, the calculated values (a=b=3.7834 Å and c=9.5181 Å) are marginally smaller in contrast to that of the standard pure anatase phase with a tetragonal structure; i.e. a=b=3.7852 Å and c=9.5139 Å (JCPDS card No. 00-021-1272). However, insignificant change in the lattice parameters is observed for the rutile phase, the lattice volume for 4 and 10% Cu-doping is around 62.42 and 62.45 Å³, respectively compared to 62.43 Å³ (a=b=4.5937 Å, c=2.9587 Å; JCPDS card No. 01-075-1750). This can be explained as follow: (i) for smaller loading, Cu²⁺ (0.73 Å) dissolve into TiO₂ host lattice by occupying Ti⁴⁺ (0.64 Å) sites resulting in lattice contraction/expansion; (ii) however, for charge balance of stoichiometric TiO₂, the compensation of the missing positive charge (+2 for each Cu²⁺ replacing one Ti⁴⁺) will be compensated by the creation of oxygen vacancy (-2); (iii) the creation of oxygen vacancy will be accompanied by a contraction of the unit cell. Therefore, the simultaneous occurrence of lattice expansion / contraction results in insignificant change in lattice parameters of TiO₂ upon Cu doping. Moreover, due to relatively large difference in ionic radii (0.64 and 0.73 Å) and charge difference (+4 and +2), a limit of solubility is reached, resulting in the precipitation of Cu metal for higher doping concentrations.

3.2 Morphological observations and chemical analysis

Figure 2 displays the surface particles' morphology of pure and 2% Cu-doped TiO₂ powders have with the corresponding elemental chemical analysis. SEM image of pure TiO₂ powder (Fig. 2a) consists of spherical-shaped particles at the nanoscale with relatively homogeneous size distribution and tendency to agglomeration. Upon 2%Cu doping (Fig. 2b), the powder consists of relatively large agglomerates of ultrafine particles (melted surfaces) with the formation of pores, resulting in improved specific surfaces and consequently very beneficial for

the photocatalytic activity. The EDX spectrum of pure TiO₂ indicates the existence of O and Ti elements only, in addition to Cu after doping, signifying the purity of the as-prepared powders. The elemental chemical composition indicates that the Ti/O ($24.2 / 50.12 = 0.482$) ratio is very close to 0.5, thereby in compliance the stoichiometry of the starting precursor solution. The amount of Cu (1.56 at. %) is also close to the intended content (2.0 at. %), in accordance with the initial precursor solution. The grain size was determined using the imageJ software. The results showed that doping with Cu decreases the particle size; from 29.23 nm for pure TiO₂ up to 19.58 nm for 2% Cu doped TiO₂, which is very beneficial for the photocatalytic reaction.

3.3 FTIR analysis

Figure 3 displays the FTIR spectra of pure and Cu-doped TiO₂ powders. All samples exhibit a broad band setting in the wavenumber range 3250 - 3700 cm⁻¹, which corresponds to the stretching vibrations of the hydroxyl groups of water molecules. The small band at 2350 cm⁻¹ for Cu-doped TiO₂ is ascribed to the CO₂ molecule in the atmosphere (Baneshi, Haghighi, Jodeiri, Abdollahifar, & Ajamein, 2014), while the band at 1635 cm⁻¹ corresponds to the bending and stretching OH groups adsorbed on the catalyst surface (Baneshi et al., 2014). Surprisingly, such bands are not detected in pure TiO₂ having, signifying that the surface of pure TiO₂ contains fewer hydroxyl groups than Cu-doped TiO₂ which can be attributed to the loss of water during calcination. In the region lower than 1071 cm⁻¹, the absorption bands are assigned to the Ti-O stretching band and O-Ti-O flexural vibration modes (Lopez et al., 2002), which confirm the formation of TiO₂ phase.

3.4 BET Measurements

The isotherm shape provides information on pore size, volume, and distribution as well as whether its macro-, meso-, or micro-porous structure. The nitrogen adsorption-desorption

isotherms are plotted for pure and 2% Cu-doped TiO₂ powders, as illustrated in Figure 4. Both curves display a sorption isotherm of type IV, which is associated with capillary condensation, respectively of nitrogen and argon adsorbed at 77 K and 87 K, occurring in mesoporous solids having cylindrical pores, as well as H1 hysteresis loops according to IUPAC classification (Sotomayor, Cychosz, & Thommes, 2018; Thommes et al., 2015). This kind of isotherm is obtained with mesoporous adsorbents having a pore radius ranging between 25 and 500 Å (Thommes et al., 2015). Such a hysteresis loop depicts uniformly arranged spherical or agglomerated nanoparticles (as shown in SEM images Fig. 2), with respective to TiO₂ higher specific surface area and pore volume (Table 2).

Surprisingly, the surface area has been dramatically reduced upon the addition of Cu, even at low content (2%), during the doping process. Systematically, a decrease of the pores' average diameter and volume is observed, ranging from (13-20 nm) to (19-24 nm) and 108×10^{-3} up to $4.5 \times 10^{-3} \text{ cm}^3 \cdot \text{g}^{-1}$, respectively. Maurice et al. (Sorolla II, Dalida, Khemthong, & Grisdanurak, 2012) found a similar effect of Cu-doping when preparing 2 wt% Cu-TiO₂ via the sol-gel route. The decline in the surface characteristics can be explained by the blocking of the pore entrance at the surface of TiO₂ (Sorolla II et al., 2012). However, the appearance of hysteresis loop in the isotherm of Cu-doped nanopowder indicates that the pore blockage does not take place over the entire material.

The BJH method has been used to determine the pore size distribution, as illustrated by the inset graphs in Figure 4. As it can be depicted in Figure 4a, the pore size distribution of pure TiO₂ nanopowder is narrow and ranging between 15 - 30 nm with an average pore diameter of about 23.4 nm. Meanwhile, the pore size distribution of Cu-doped nanopowder is dispersed in a broad interval from 10 to 40 nm with an average pore diameter equal to 19.6 nm (Fig. 4b). As it can be

observed, the incorporation of Cu within TiO₂ host lattice even at such low doping level generated lowering of pore diameter and volume as well as broadening and uniformity of pore size distribution. This finding implies that Cu is well dispersed into TiO₂ network causing some modifications of the mesostructure at its surface, which corroborate with XRD analysis.

3.4. Photodegradation kinetics study of Azucryl red

The study of the photodegradation kinetics and mechanism is very essential in processing aqueous wastewater as it avail precious information on the degradation process. The degradation kinetics has been performed by varying the amount of catalyst, the initial pH of AR solution as well as the initial AR dye concentrations. Equation (6) represents a pseudo-first-order model of Lagergren (Shahwan, 2015; Sorolla II et al., 2012) with a rate constant k_{app} (min⁻¹) and q_e the maximum quantity of AR adsorbed at equilibrium. The plot of C_t/C_0 against t allows the determination of k_{deg} :

$$\frac{C_t}{C_0} = \text{Exp}(-k_{deg} \cdot t) \quad (7)$$

The kinetics results of the photodegradation process under UV irradiation are displayed in Figure 5 (a, b, and c). The values of k_{deg} and the linear regression coefficient (r^2) are computed by the simulation of the model using Origin Pro software and are summarized in Tables 3. According to the regression coefficient, mainly greater than 0.96, the experimental results are well simulated by the Lagergren pseudo-first-order model for the tested adsorbents. As can be noted, from Figure 5, photolysis does not affect the degradation of AR; however, the presence of the photocatalyst reduces extremely the concentration of the dye. Based on the obtained rate constants, the half-life time ($t_{1/2} = \ln 2/k_{deg}$) has been also calculated and compared (Shahwan, 2015). As shown in Table 3, the value of $t_{1/2}$ obtained based on the rate constant of Eq. (6) ranges from 18 up to 65 min, which is much closer to the experimental values.

The adsorption of AR on the surface of the catalyst and its photodegradation can be governed by the mass transfer through the boundary liquid film and/or by the intra-particle diffusion/transport process. The linearized equation of mass transfer kinetic model proposed by Ketcha Mbadcam *et al.* (Ho & McKay, 1998) is expressed as follows:

$$\ln(C_0 - C_t) = \ln(D) + k_0 \cdot t \quad (8)$$

where D is the mass transfer constant and k_0 (min^{-1}) the adsorption constant. AR may be carried from the liquid phase to the solid particles by the intra-particle diffusion process. The mass transfer kinetic model constants are obtained from the slope and the intercept of the straight line $\ln(C_0 - C_t)$ vs. time. The linear plots illustrated in Figure 6 (a, b, and c) and mass transfer model constants are given in Table 3.

Sometimes, such a mechanism model is a rate-limiting stage in the adsorption or photodegradation process. The probability of intra-particle diffusion is examined through the Weber and Morris diffusion model (Hameed, Salman, & Ahmad, 2009; Mbadcam, Anagho, & Nsami, 2011):

$$q_t = k_{dif} \cdot t^{1/2} + C \quad (9)$$

where C , the intercept yields more details regarding the thickness of the boundary layer. The intra-particle diffusion constant, k_{dif} value (in $\text{mg} \cdot \text{g}^{-1} \cdot \text{min}^{1/2}$), for the tested adsorbent is obtained from the slope of the graph (Fig. 7 (a, b and c)) as illustrated in Table 3. The validity of these models is thereafter discussed on the basis of the regression coefficient r^2 .

3.4.1 Effect of Cu-doping concentration

Figure 5a displays the photocatalytic decomposition of the azucrylic dye in aqueous solution by pure and Cu-doped TiO_2 nanopowders under UV irradiation. As it can see, the azucryl red decays constantly throughout the irradiation time for all nanopowders. The photocatalytic

elimination efficiency of Cu-doped TiO₂ for 2% and 4% reaches approximately 85.26 and 77.64% of the initial concentration after 60 minutes of irradiation, respectively. Whereas beyond this value, the degradation rate decreases, and pure TiO₂ shows only 73.84% of AR elimination efficiency. This indicates that Cu-doping even in small loading, promotes the photocatalytic reaction. Therefore, 2%Cu represents the optimal metal charge for the effectiveness of strengthening the photocatalytic activity of TiO₂.

From Table 3, it can be observed that the highest constant efficiency (K_{deg}) is obtained for 2 and 4% Cu, with 3.83×10^{-2} and $2.7510^{-2} \text{ min}^{-1}$, respectively. Then it decreases drastically to $1.16 \times 10^{-2} \text{ min}^{-1}$ for 10% Cu beneath the obtained value for pure TiO₂. This finding may indicate that a larger content of copper will deteriorate the catalytic photodegradation capacity.

There may be the probable causes for the reduction of the photocatalytic efficiency at high Cu-doping content: (i) the active sites present at the surface of catalyst are covered with an excess of Cu; (ii) the combination of the charge carriers increases the rate photocatalytic degradation with higher amount of Cu charge (Kim et al., 2013; Morikawa, Irokawa, & Ohwaki, 2006).

3.4.2 Effect of initial AR solution pH

The influence of the initial solution pH (i.e., pH = 3.0, 5.0, 6.0, and 8.0 ± 0.1) on the degradation of AR by pure and Cu-doped TiO₂ nanopowders under UV-irradiation, are presented in Figure 5b and the calculated Lagergren model constant values are given in Table 3.

The obtained results show that the azucryl red photodecoloration is higher in the case of acidic pH medium, and the value of K_{deg} constant increases for the pH change from 3.0 to 5.0, subsequently decreases for higher pH value. The most significant degradation kinetics is observed at a pH of 5.0 (Table 3). Besides, the effect of the solution's initial pH on the photodegradation process is

substantial, since it influences the electric charge of the particles. Indeed, in acidic medium, the dye adsorption occurs at the surface of TiO₂ particles, may be due to the electrostatic attraction between the positively charged TiO₂ and the negatively charged dye. However, the rate of photodegradation decreases with increasing pH (pH > 6). In the present study, the azucryl red is deprotonated and negatively charged for the basic pH. The same applies to TiO₂ (Katz, McDonagh, Tijing, & Shon, 2015), and the repulsion between the azucryl red anion and TiO₂ particles decreases the initial velocity of the dye pollutant degradation.

3.4.3 Effect of AR initial concentration

Figure 5c depicts the influence of the initial Azucryl red concentration (5, 10, 15, and 20 mg. L⁻¹) on the photocatalytic degradation process in the presence of 0.2 mg of pure and Cu-doped TiO₂ catalysts. The Lagergren constant and r² values are summarized in Table 3. The obtained results prove that the effect of the initial concentration of Azucryl red is inversely proportional to the rate of degradation. Indeed, the higher the concentration of Azucryl red, the longer the degradation time is, until 15%Cu doping concentration. Beyond this value, the effectiveness of AR degradation has increased, which can be explained by the enhancement in the pollutant-surface attraction to the photocatalyst. Thus, for the same period of irradiation, i.e., 60 min, the degradation yield of Azucryl red in solutions of 5 and 10 mg. L⁻¹ is found to be 85.2 and 57.55%, respectively.

In contrast, for higher concentration of 15 and 20 mg. L⁻¹, the degradation yield reaches a steady state; i.e. only 45.96 and 46.89%. Several authors reported this phenomenon related to the efficiency of photocatalytic degradation and the decrease in constant velocity with increasing substrate concentration (da Silva & Faria, 2003). This result can be explained by the fact that when

the dye concentrated in a solution is high; the phenomenon of diffusion of light by the dye molecules will be essential and will have a negative influence on the degradation activity.

The photodegradation of AR at the surface of the tested adsorbents may be controlled by the intra-particle rate model, since, q_t and $t^{1/2}$ convene a linear correlation (Fig. 7 (a, b, and c)). Besides, the regression coefficient values, regardless of the studied conditions, are mostly higher than 0.96, signifying that the data are strictly appropriate to the model. The thickness of the boundary layer is highly matched up with the C values. Nevertheless, Figure 7 (a, b, and c), linear curves do not pass across the origin. This annotation can be ascribed to some level of boundary layer control. The mentioned phenomenon is an allusion that the intra-particle diffusion is not the only rate-limiting stage, as supplementary kinetic processes may affect the photodegradation activity. In other words, the as mentioned physic-chemical parameters are intervening concurrently and cannot be neglected (Royer et al., 2009; Yazdani, Tuutijärvi, Bhatnagar, & Vahala, 2016). Meanwhile, this finding can be correlated by the results obtained using mass transfer model, namely, the high values of regression coefficients (Tables 3). Besides, linear relation has been observed (Fig. 6 (a, b, and c)), and the regression coefficients are higher than 0.91, except in the case of Cu-doping concentration 10% and pH 6.2 (0.7966, 0.8324, respectively). This finding points out that the photodegradation kinetic process may also be controlled by mass transfer through a liquid-particle interface, namely, the mass transfer by convection (Royer et al., 2009).

3.5. Photodegradation Equilibrium study of azucryl red

Photodegradation isotherms can represent the equilibrium established between the azucryl red on the adsorbent and the unadsorbed component in the solution. The most widely used isotherm equations for equilibrium data modeling are Langmuir (Langmuir, 1918), Freundlich (Freundlich,

1907), Temkin (Allen, Mckay, & Porter, 2004), and Dubinin-Radushkevich (D-R) (da Silva & Faria, 2003) isotherms and they are expressed as follow:

$$\frac{C_e}{q_e} = \frac{1}{q_m} C_e + \frac{1}{q_m \cdot b} \quad \text{linear form of Langmuir equation} \quad (10)$$

$$\text{Ln}q_e = \frac{1}{n} \text{Ln}C_e + \text{Ln}K \quad \text{linear form of Freundlich equation} \quad (11)$$

$$q_e = \frac{RT}{B} \text{Ln}K_T + \frac{RT}{B} \text{Ln}C_e \quad \text{linear form of Temkin equation} \quad (12)$$

$$\text{Ln}q_e = \text{Ln}q_m - \beta \varepsilon^2 \quad \text{linear form of D-R equation} \quad (13)$$

$$\text{with } \varepsilon = RT \text{Ln}(1 + \frac{1}{C_e})$$

where q_m is the concentration of sorbed adsorbate at nanoparticle-phase, corresponding to the complete monolayer coverage of adsorption sites (Langmuir, 1918), and b is the Langmuir constant concerning the adsorption free energy. The q_m and b values are evaluated from the Langmuir model linear plot (Eq. 9): the slope is $1/q_m$ and the intercept $1/(q_m \cdot b)$. The Freundlich model's constants, k and n , are determined from the slope and intercept of the plot $\text{Ln}(q_e)$ against $\text{Ln}(C_e)$ (Eq. 10), respectively. K and n constants may be correlated to the strength of the adsorptive bond and the bond distribution, respectively (Freundlich, 1907). The Temkin's isotherm presumes that the heat of adsorption of all molecules increases linearly with coverage (Allen et al., 2004). The linear form of this isotherm can be given by Eq. (11), where B and K_T are Temkin constants, respectively related to the heat of sorption ($\text{J} \cdot \text{mol}^{-1}$) and Temkin's isotherm constant ($\text{L} \cdot \text{mg}^{-1}$). The constants B and K_T are determined from the slope and the intercept of the linear plot of q_e against $\text{Ln}(C_e)$. Dubinin-Radushkevich isotherm expression is represented by the equation (12) (Ayawei, Ekubo, Wankasi, & Dikio, 2015), where ε is Polanyi potential, β is D-R's constant, R is gas constant ($8.31 \text{ J} \cdot \text{mol}^{-1} \cdot \text{K}^{-1}$), T is absolute temperature, and $E = (2\beta)^{-0.5}$ is mean adsorption energy.

It is well-known that the adsorption isotherms help in describing the degradation process at equilibrium conditions. Figure 8 depicts the photodegradation equilibrium isotherms of AR dye (5, 10, 15, 20, and 25 mg.L⁻¹) by Cu₂%-TiO₂ nanopowder for 60 minutes at room temperature and 5.0 ± 0.2 pH. Besides, the batch experiments data were modeled by the four previously described isotherm models using the least-squares method. The linearized Langmuir, Freundlich, Temkin and D-R isotherms of AR dye in solutions are plotted. Consequently, the respective model parameters (q_m , b , K , n , B , KT , ε , and β) with a correlation coefficient (r^2) are determined by the simulation of the different models by using Origin 8 software and summarized in Table 4.

The substantial characteristic of the Langmuir model may be manifested in the expression of equilibrium parameter R_L , which is a constant known as the separation factor (Langmuir, 1918):

$$R_L = \frac{1}{1 + b.C_0} \quad (14)$$

where R_L value indicates the adsorption nature as irreversible if $R_L = 0$, favorable if $0 < R_L < 1$, linear if $R_L = 1$ and unfavorable if $R_L > 1$.

Taking into account the high values of the regression coefficient ($r^2=0.9935$) as shown in Table 4, it should be pointed out that the Freundlich model exhibits a better fitting for AR dye photodegradation equilibrium data (Fig. 8 and Table 4). Based on this finding, it can be highlighted that the photodegradation data are better described by Freundlich isotherm compared to the remaining models' isotherms, particularly Langmuir. However, from the obtained results, the maximum monolayer coverage capacity (q_m) from the Langmuir isotherm model has been found to be 120 mg.g⁻¹ and R_L is about 0.03, indicating that the equilibrium photodegradation is explicitly more favorable ($r^2 = 0.9651$).

3.6 Photocatalytic degradation mechanism

A plausible mechanism for the photocatalytic degradation of AR dye is illustrated in Fig. 9. By means of illumination of the as-prepared $\text{Cu}_x\%-\text{TiO}_2$ nanocatalysts (NC) with UV-light; the electrons in the valence band (VB) are excited to the conduction bands (CB), consequently an equivalent number of holes are formed in the VB.

By means of radiation, the as-prepared $\text{Cu}_x\%-\text{TiO}_2$ nanocatalysts (NC) with UV-light wavelength greater than or equal to its bandgap, the electrons move from the valence band (VB) to the conduction band (CB) forming an equivalent number of an oxidizing site (hole h^+) and a reduction site (electron e^-) (Ajmal, Majeed, Malik, Idriss, & Nadeem, 2014). Meanwhile, TiO_2 is a semiconductor with a wide bandgap. Furthermore, the incorporation of Cu^{2+} into the TiO_2 matrix will produce intermediate energy levels that occasionally retard in the charge carrier's recombination and, consequently, reinforce the photocatalytic efficiency. The latter process is the oxidation-reduction reaction of the adsorbed reagents on the NC particle surface (Ajmal et al., 2014), the charges arriving at the photocatalyst surface react with the adsorbed substances that can accept or donate electrons. The electrons react with oxygen to yield the superoxide radicals (Kaur & Singhal, 2014), which may interact with the cations H^+ to form the radical hydro-peroxide HO_2 : (i) and the formation of hydrogen peroxide that decomposes HO^* hydroxide radical (ii) to the catalyst surface after excitation by lengths appropriate wave. The h^+ holes react with the reagents adsorbed on the NC surface liable to give electrons such as water, adsorbed OH^- anions or directly organic pollutants (AR), producing highly oxidizing hydroxyl radicals which can oxidize the organic molecules adsorbed on TiO_2 surface until complete mineralization into CO_2 and H_2O as shown in Figure 9.

3.7 Regeneration of the adsorbent

The reusability of photocatalysts nanocomposite is required for their feasible utilization on a large scale for industrial applications. The procedure used to remove dyes molecules from wastewater systems is considered upon numerous criteria like cost, efficiency, and reusability. The recycling of adsorbents and the renewal of adsorbate is a cost-effective process. In this context, the regeneration of the 2% Cu-doped TiO₂ nanoparticles from the dye mixture has been investigated for five cycles. After the photodegradation process, the nanocatalyst was only thoroughly washed with deionized hot water and oven-dried at 100 °C for three hours. The measurements are realized similarly to the photodegradation kinetics and equilibrium process, i.e., pH = 5.0 during 60 minutes of contact time. The adsorption–desorption process has been repeated up to five cycles to test the reusability of the nanoparticles for AR dye molecules degradation. As depicted in Fig. 10, 2% Cu-doped TiO₂ nano-photocatalyst indiscernibly displays a higher degree of stability, and its efficiency is found to be relatively significant for AR dye after five sequential cycles. It is also important to highlight that 2% Cu-TiO₂ nanoparticles fabricated via an amended sol-gel method can be recovered effectively for the reuse in photodegradation for more than five cycles.

4. Conclusion

TiO₂ and Cu-TiO₂ nanopowders were synthesized, characterized, and tested as catalysts in the azucryl red dye's photodegradation. XRD data indicated a rutile-anatase phase transformation for 2% Cu doping, while the crystallite size was slightly affected by Cu loading (around 30 nm). The morphology of the nanoparticles has transformed from spherical-shaped into agglomerated large nanoparticles upon Cu doping. The BET analysis revealed a sudden drop in surface area from 16.5 m².g⁻¹ for TiO₂ to about 1.0 m².g⁻¹ after 2% Cu-doping content addition. Photocatalytic

efficiency up to 86 % was achieved at 2% Cu-doping content, at 25°C and a pH of 5.0. The kinetics study of the dye photodegradation process was proved to obey the Lagergren first-order law, regardless of experimental conditions (i.e., %Cu doping concentration, pH, and initial dye concentration). The kinetic investigation indicated that the rate is not only controlled by the intra-particle diffusion process, particularly at its initial phase. Instead, it was proved that this process might also be controlled by mass transfer through the liquid phase's boundary layer. Besides, the Freundlich isotherm yielded a better fitting of the equilibrium data of the photodegradation process. Apart from being highly reusable, the photocatalyst showed excellent performance at a low Cu concentration Cu also. The as-obtained findings on Cu-doped TiO₂ nanopowder demonstrated it as a potential nanocatalyst contender for organic dyes photodegradation.

Authors' contributions:

Ahmed Kerrami: validation, investigation, visualization, resources, writing –original draft, writing - review & editing.

Lotfi Khezami: methodology, validation, writing –original draft, writing - review & editing. validation, investigation.

Mohamed Bououdina: conceptualization, validation, writing –original draft, writing - review & editing, visualization, supervision.

Laila Mahtout: conceptualization, methodology, resources, supervision.

Abueliz Modwi: methodology, validation, investigation, review & editing.

Souhila Rabhi: validation, investigation, review & editing.

Faycal Bensouici, Hayet Belkacemi: conceptualization, methodology, resources, supervision.

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Figure Captions

Fig. 1 X-ray diffraction patterns of copper-doped TiO₂ nanopowders.

Fig. 2 SEM images with corresponding EDS spectra of (a) pure TiO₂, and (b) 2% Cu-doped TiO₂ nanopowders.

Fig. 3 Infrared spectra of undoped and Cu-doped TiO₂ nanopowders.

Fig. 4 N₂ adsorption-desorption Isotherm of (a) pure TiO₂, (b) 2%Cu-TiO₂ nanopowders and inset figures are BJH pore size distribution of pure TiO₂, 2%Cu-TiO₂ catalysts.

Fig. 5 Kinetics UV photocatalytic degradation at different conditions of (a) Cu-doping loading; (b) concentration initial pH, and (c) initial concentration of AR.

Fig. 6 (a) Mass transfer model plots for the photodegradation of 0.5 mg.L⁻¹ AR by Cu_x%-TiO₂ nanopowder at pH 5.0; (b) Mass transfer model plots for the photodegradation of AR by Cu₂%-TiO₂ nanopowder at different solution initial pH; (c) Mass transfer model plots for the photodegradation of different initial doses of AR by Cu₂%-TiO₂ nanopowder at pH 5.0.

Fig. 7 (a) Intra-particle diffusion plots for photodegradation of 0.5 mg.L⁻¹ AR by Cu_x%-TiO₂ nanopowder at pH 5.0; (b) Intra-particle diffusion plots for photodegradation of AR by Cu₂%-TiO₂ nanopowder at different solution initial pH; (c) Intra-particle diffusion plots for photodegradation of different initial doses of AR by Cu₂%-TiO₂ nanopowder at pH 5.0.

Fig. 8 photodegradation isotherm models for AR dye by Cu-doped TiO₂ nanopowders using batch mode procedure at 25°C, adsorbent mass of 0.2 mg; pH fixed at 5.0; and using a contact time of 60 min. Comparison of isotherm models non-linear plot.

Fig. 9 Schematic AR photodegradation over Cu-doped TiO₂ nanocatalysts.

Fig. 10 Reusability of 2% Cu-doped TiO₂ photocatalyst for five successive cycles for the degradation of AR.

658 **Table captions**

659 **Table 1.** Structural and microstructural parameters of pure and 2% Cu-doped TiO₂ nanopowders
660 as obtained from XRD analysis.

661 **Table 2.** BET characteristics of pure and 2% Cu-doped TiO₂ nanopowders.

662 **Table 3.** Intraparticle diffusion, mass transfer, and Lagergren pseudo-first-order models'
663 constants for the photodegradation of AR by pure and Cu-doped TiO₂ nanopowders at different
664 conditions.

665 **Table 4.** Different equilibrium model isotherm constants for AR photodegradation by 2% Cu-
666 doped TiO₂ nanopowder [m = 0.2mg, pH = 5.0, T = 25°C].

Figures

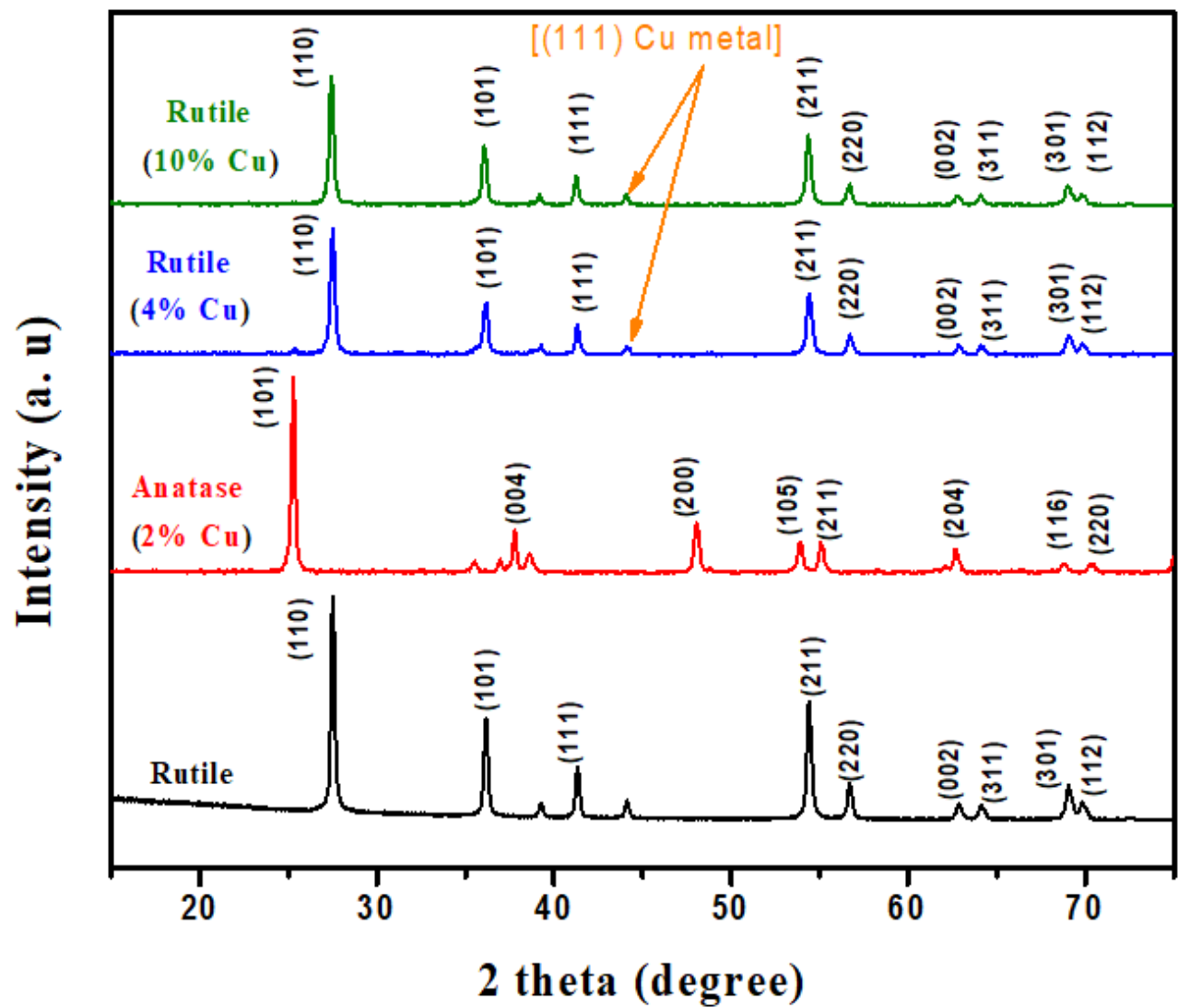


Figure 1

X-ray diffraction patterns of copper-doped TiO₂ nanopowders.

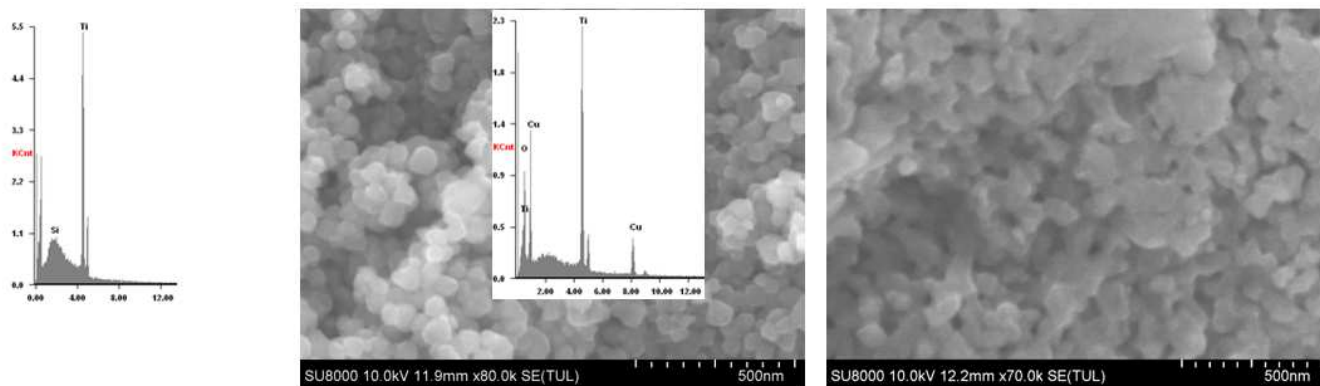


Figure 2

SEM images with corresponding EDS spectra of (a) pure TiO_2 , and (b) 2% Cu-doped TiO_2 nanopowders.

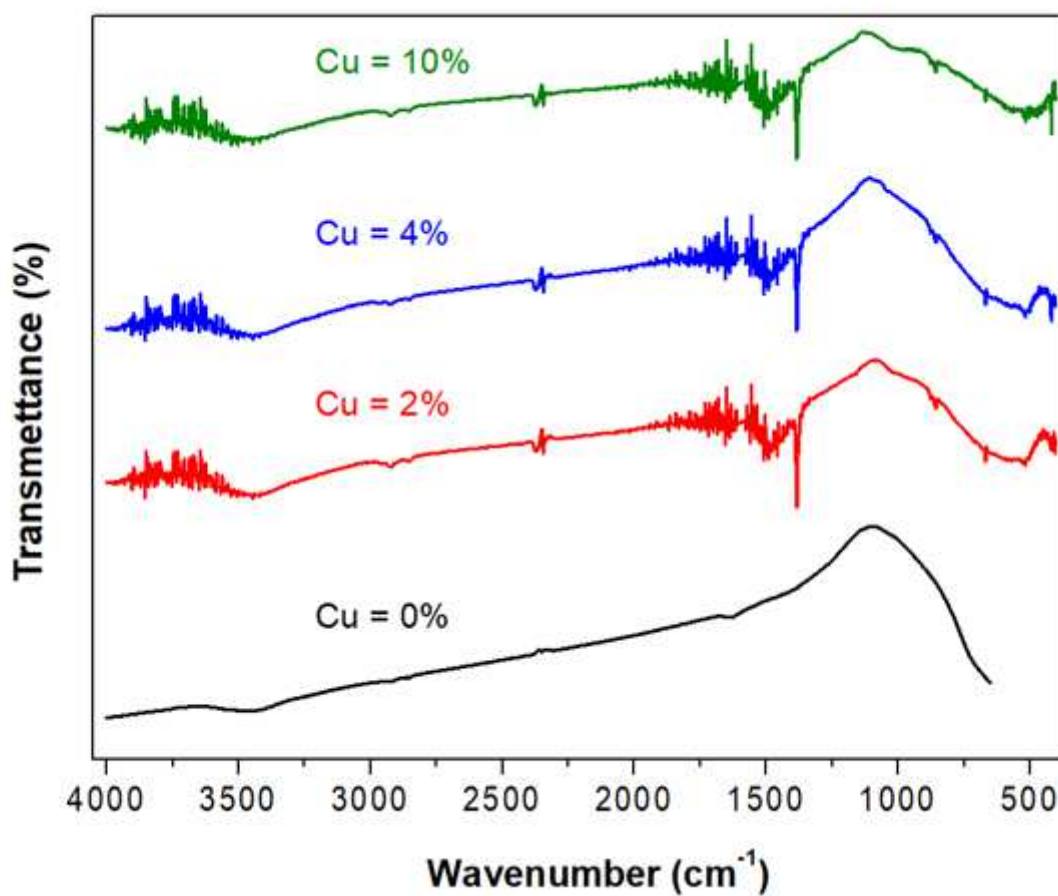


Figure 3

Infrared spectra of undoped and Cu-doped TiO_2 nanopowders.

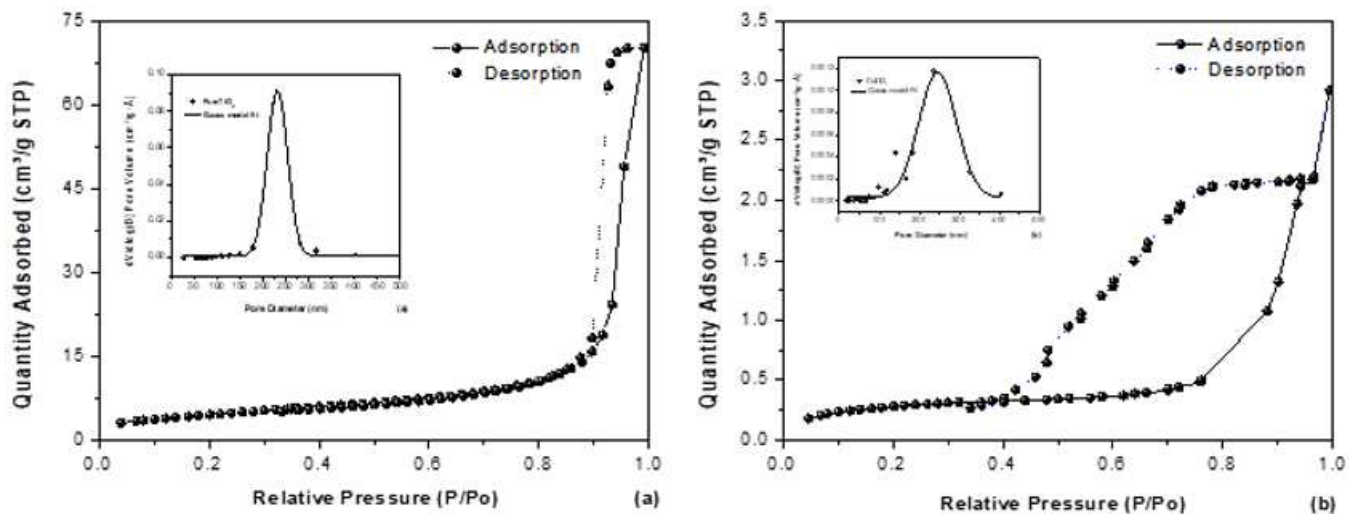


Figure 4

N₂ adsorption-desorption Isotherm of (a) pure TiO₂, (b) 2% Cu-TiO₂ nanopowders and inset figures are BJH pore size distribution of pure TiO₂, 2% Cu-TiO₂ catalysts.

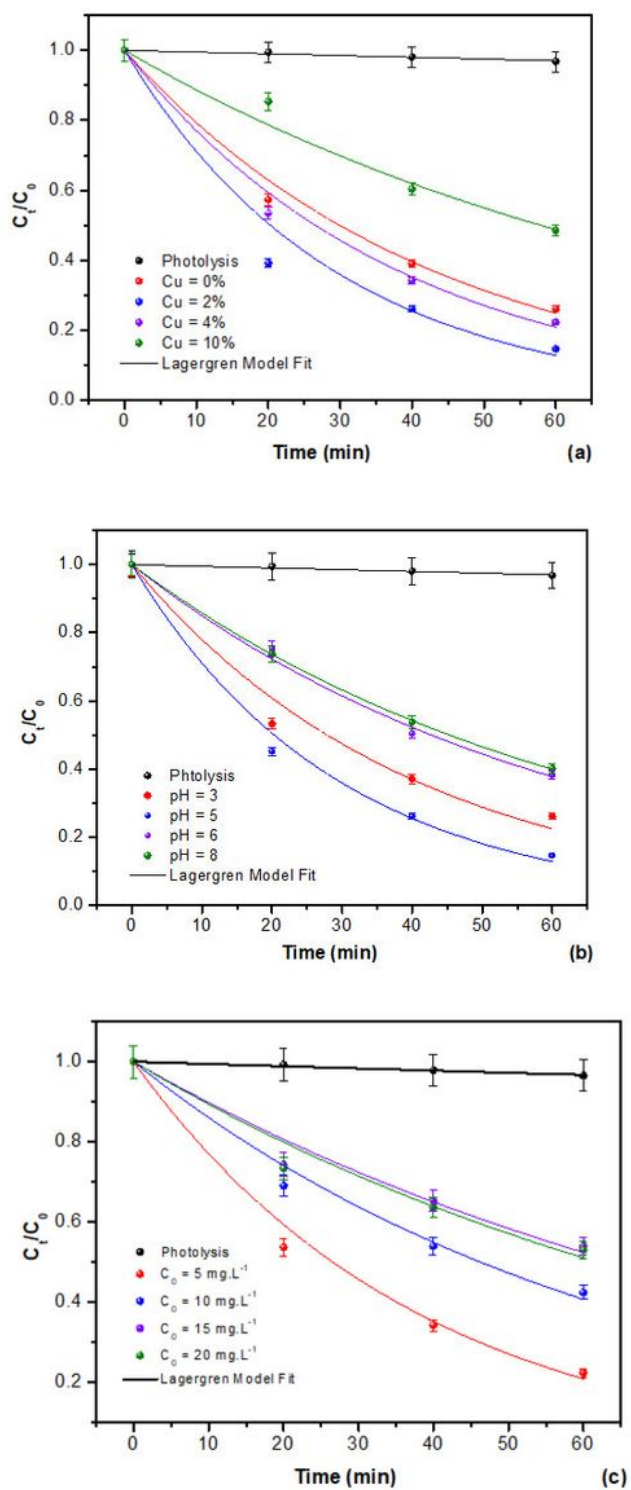


Figure 5

Kinetics UV photocatalytic degradation at different conditions, of (a) Cu-doping loading, (b) concentration initial pH, and (c) initial concentration of Ar

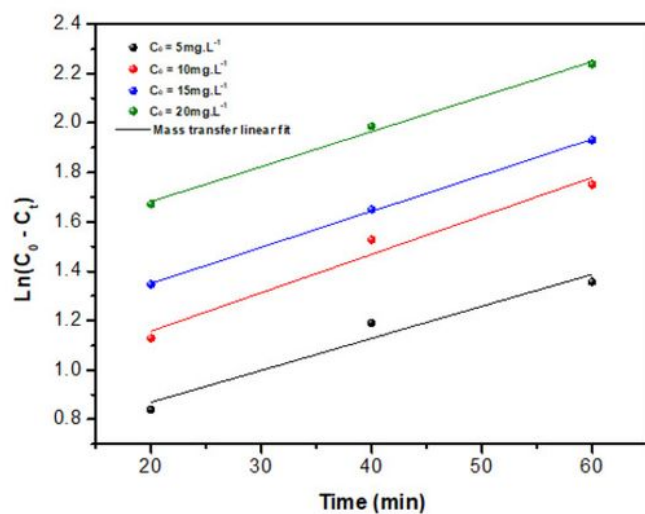
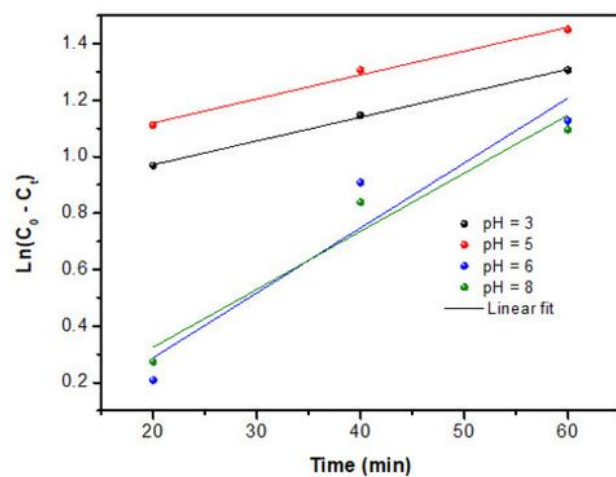
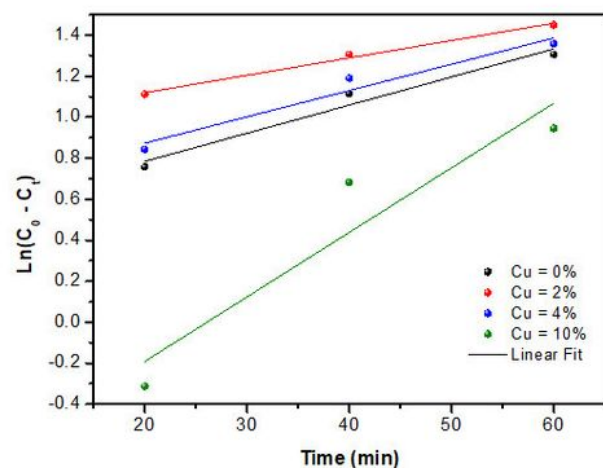


Figure 6

Top panel-a Mass transfer model plots for the photodegradation of 0.5 mg.L⁻¹ AR by Cux%-TiO₂ nanopowder at pH 5.0. Middle panel-b Mass transfer model plots for the photodegradation of AR by Cu₂%-TiO₂ nanopowder at different solution initial pH. Bottom panel-c Mass transfer model plots for the photodegradation of different initial doses of AR by Cu₂%-TiO₂ nanopowder at pH 5.2

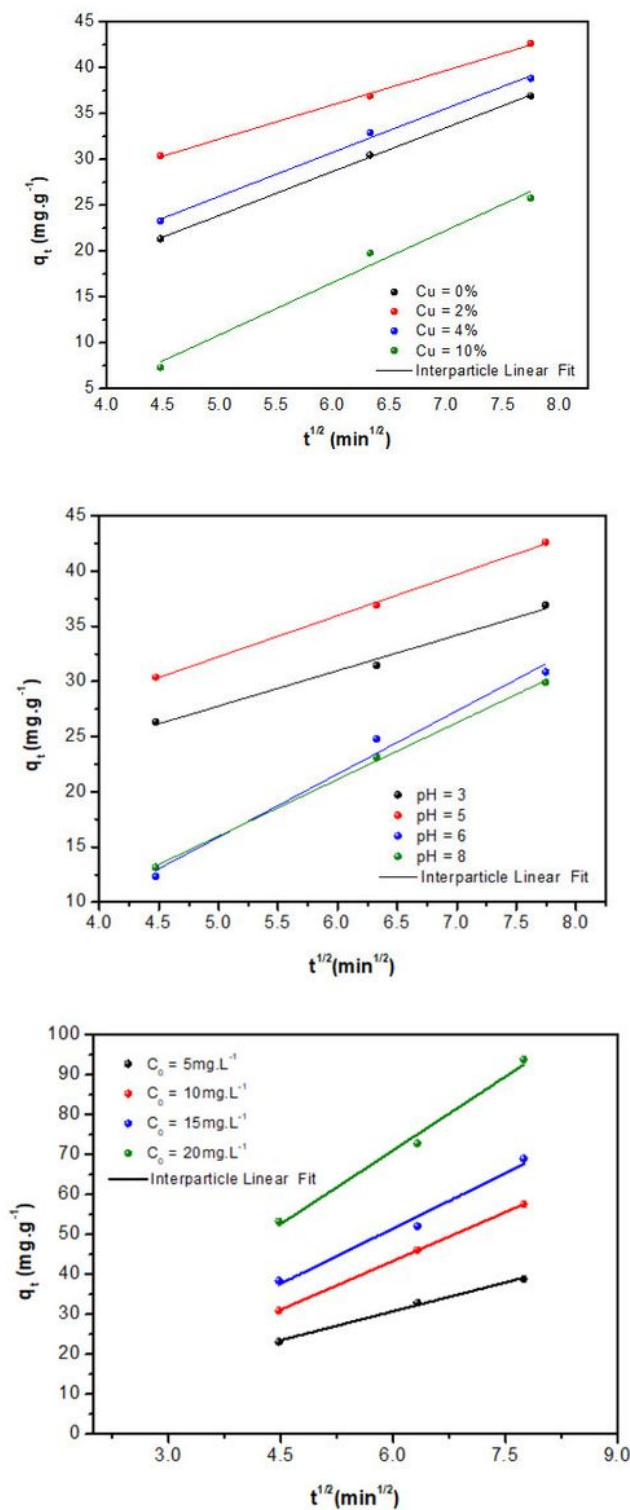


Figure 7

Top panel-a Intra-particle diffusion plots for photodegradation of 0.5 mg.L⁻¹ AR by Cu_x%-TiO₂ nanopowder at pH 5.0. Middle panel-b Intra-particle diffusion plots for photodegradation of AR by Cu₂%-TiO₂ nanopowder at different solution initial pH. Bottom panel-c Intra-particle diffusion plots for photodegradation of different initial doses of AR by Cu₂%-TiO₂ nanopowder at pH 5.0.

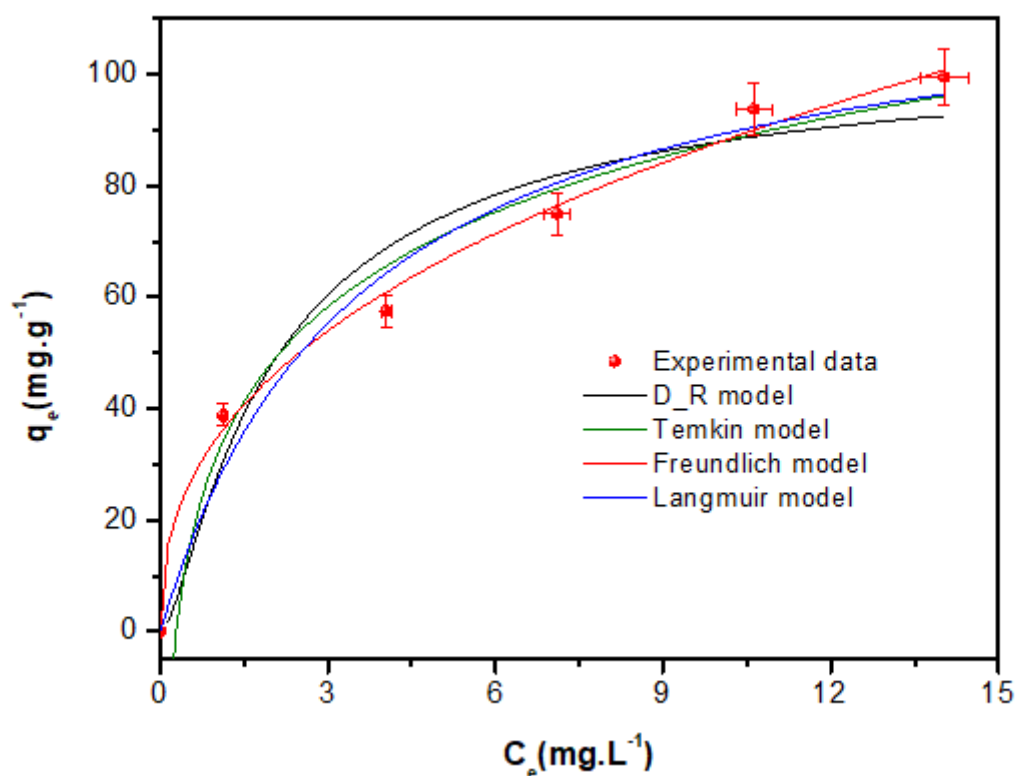


Figure 8

Photodegradation isotherm models for AR dye by Cu-doped TiO₂ nanopowders using batch mode procedure at 25°C, adsorbent mass of 0.2 mg; pH fixed at 5.0; and using a contact time of 60 min. Comparison of isotherm models non-linear plot.

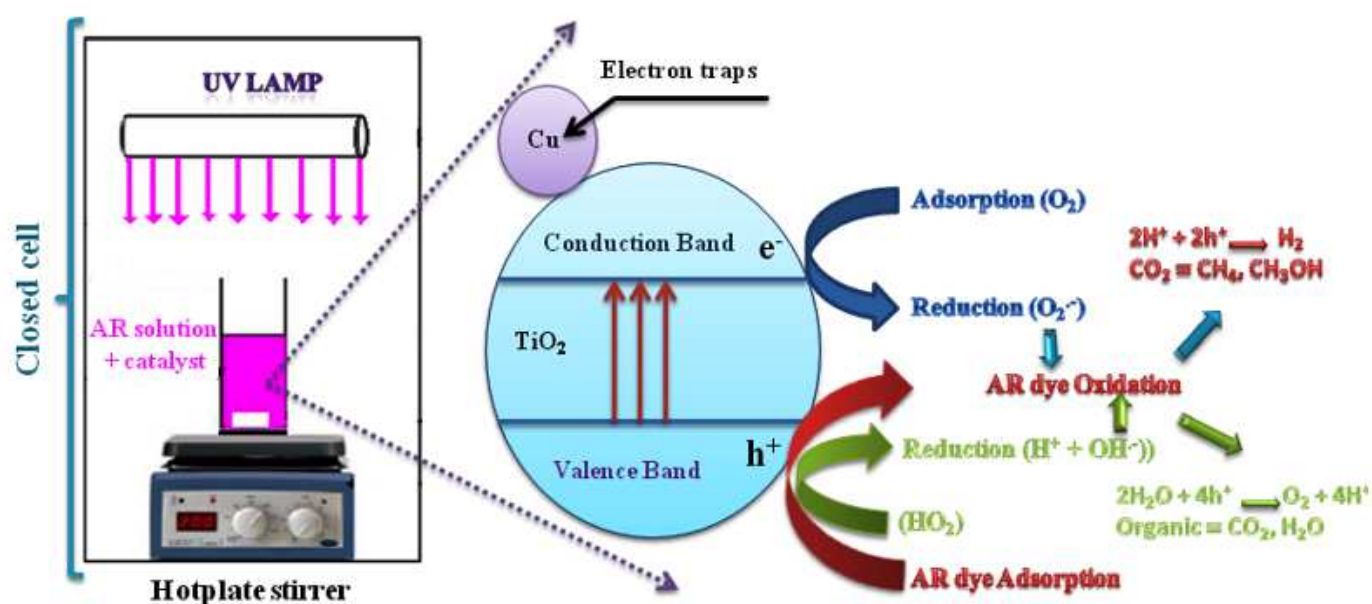


Figure 9

Schematic AR photodegradation over Cu-doped TiO₂ nanocatalysts.

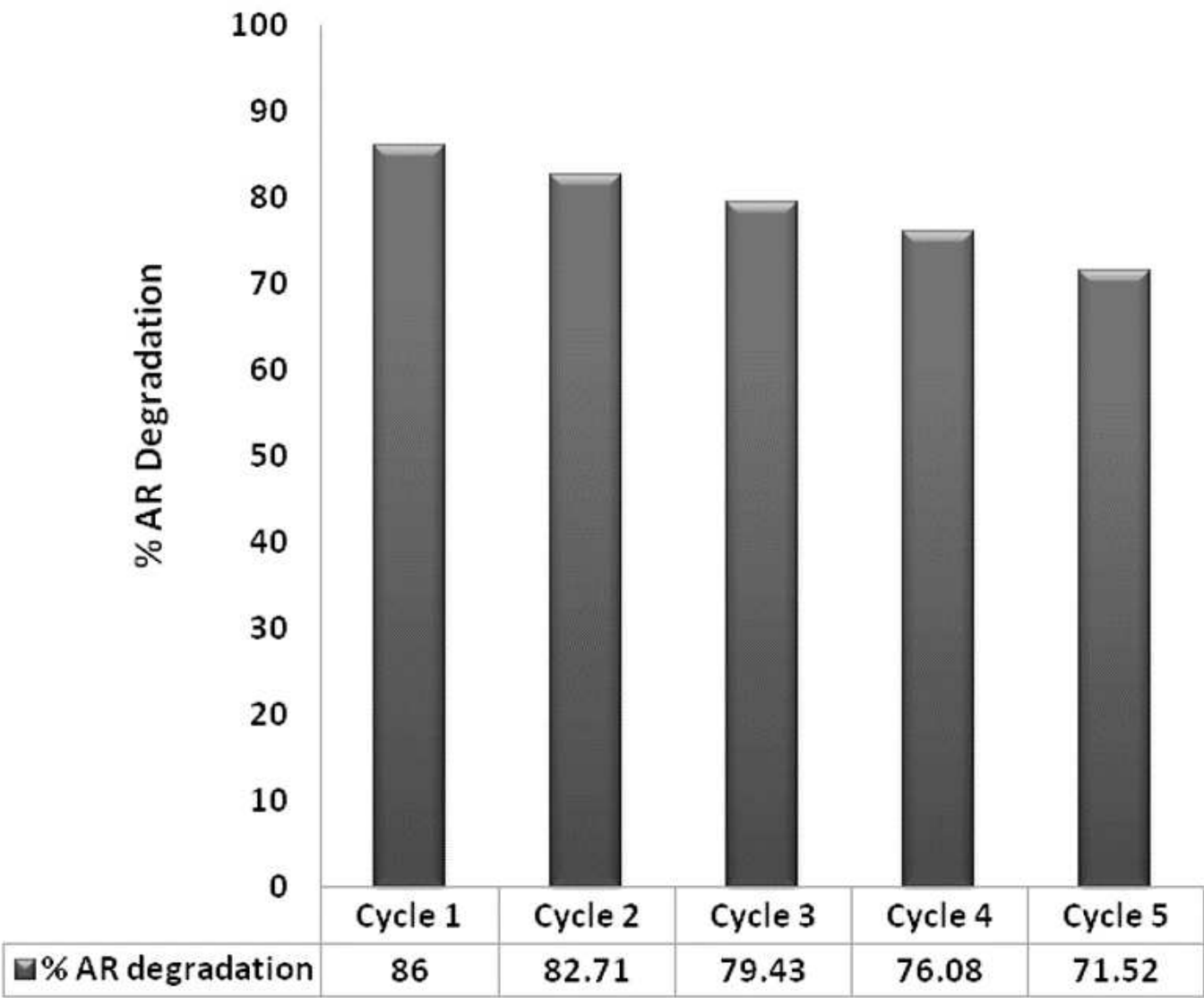


Figure 10

Reusability of 2% Cu-doped TiO₂ photocatalyst for five successive cycles for the degradation of AR.