

Control of hysteretic sorption of flexible MOF for practical acetylene safe storage

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ABSTRACT. Safe managing and storage of explosive gas, such as acetylene, is essential for industries. The practical pressure and temperature regime is quite narrow for acetylene, and the optimization by complete release of saturated gas remains challenging to control. Here, a novel approach to narrow the pressure range of gas introduction and release for the storage of unstable and pressure-sensitive gas substance is reported. Flexible metal–organic frameworks (MOFs) are expected to be an excellent storage adsorbent owing to their gated adsorption/desorption behavior. However, developing such a system in a practical and narrow pressure range remains difficult. Here, we demonstrate that the gate-opening and closing pressures of a solid solution type based on interpenetrated MOF-508 can be optimized by tuning the deformation energy of the framework and the energy of host–guest interactions. These energies were adjusted by changing the functional

group on the MOF ligands and its ratio in the mixed-ligand system. The filling and release of acetylene in the practical pressure regions of up to 180 kPa leads to an on-demand porous material for acetylene storage with an usable volumetric capacity of 106 v/v under mild temperature (273–298 K) and pressure (70–120 kPa).

The development of new technologies for efficient gas storage at ambient pressure can provide improved safety and new possibilities in terms of gas cylinder design for gas supply industries. The usable gas capacity under mild pressure and temperature conditions, which often differs from the maximum quantity adsorbed, is another concern for researchers. As a potential solution, porous materials and, in particular, metal–organic frameworks (MOFs) or porous coordination polymers (PCPs) possess inherent voids, which allow to store and deliver large amounts of gases.¹⁻³ However, rigid MOF materials, with a Langmuir type I isotherm, exhibit close adsorption and desorption pressures, which result in a low usable capacity (**Fig. 1**). In contrast, flexible-type MOFs, which are called soft porous crystals (SPCs), possess both ordered network and structural transformability, which provide them higher usable capacity.^{4,5} Indeed, the flexibility of SPCs allows to distinguish filling from release pressures, which results in an isotherm with a sigmoidal shape characterized by a hysteresis (Δ) between the gate-opening pressure (P_{go}) and desorption completion (gate closing) pressure (P_{gc}), with $\Delta = P_{go} - P_{gc}$ (**Fig. 1**).⁶

Although many SPCs have been synthesized and S-shaped adsorption curves have been reported, the main focus was the discovery and understanding of phenomena rather than their practical applications. Indeed, to control the adsorption properties of SPCs, different strategies have been reported to tune the parameters responsible for structural flexibility.^{7,8} The latter depends not only on the binding capacity and mobility of unit ligand and metal ion but also on factors such as the

deformation of the entire framework owing to the accommodation of guest molecules in the pores.⁹ Even if strategies using ligand functionalization have been reported to investigate the properties of flexible MOFs, their use for the rational control over the gate-opening and closing pressure remains challenging.¹⁰⁻¹³ Therefore, a novel approach for gas storage is required with a narrow pressure range of gas introduction and release and with a low pressure limit for pressure-sensitive unstable gas substance.

Here, we chose $[\text{Zn}_2(\text{bdc})_2(4,4'\text{-bpy})]$ (bdc = 1,4-benzenedicarboxylic acid; 4,4'-bipyridine) (MOF-508) as a candidate owing to its relative displacement of mutually interpenetrating frameworks and overall crystal softness. The gate opening and closing, which results in acetylene sorption, was successfully tuned for storage application. The introduction of a different ratio of functionalized **bdc-X** linkers (**X** = $-\text{NO}_2$ or $-\text{NH}_2$) modulates the uptake and release pressures of various gases at different temperatures while keeping a similar uptake capacity. The variation in the gate-opening pressure is explained in detail, accurately predicted within isostructural mixed-ligand MOFs, and is attributed to the interplay between framework–framework and framework–guest interactions. To understand the outcome of affinity and steric effect of the ligand substituent, *in-situ* PXRD measurements combined with synchrotron powder X-ray diffractions and thermodynamic calculations were performed to provide insight on the mechanism behind the gate-opening pressure modulation. The ability to finely tune the loading and release pressures was demonstrated on acetylene as a case study of direct applicability for the industry. Indeed, owing to its explosive nature, acetylene is typically stored in an adsorbent filler combined with a solvent to improve safety (pressure limit of 2 bar).¹⁴⁻¹⁶ Creating a material that can store enough acetylene in this very narrow pressure range is challenging (70–160 kPa). Up to now, only a few studies on acetylene gas storage using SPC materials have been

conducted. This gas uptake and release pressure control strategy is applicable for well-adapted gas storage containers.

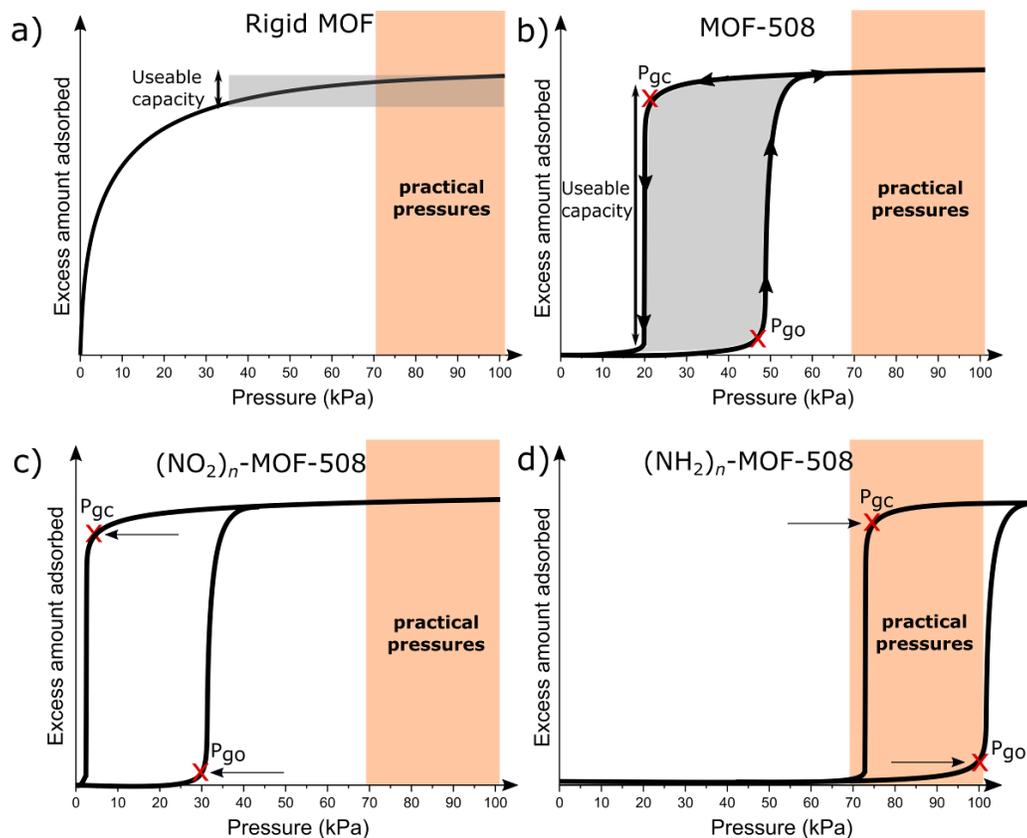


Fig. 1 | Comparison of acetylene gas sorption profiles and usable capacity in the pressure range of 0–100 kPa for **a)** a rigid material showing a type I Langmuir isotherm, **b)** flexible interdigitated **MOF-508** showing a S-shape isotherm with a gate-opening pressure (P_{go}) and a gate closing pressure (P_{gc}). The orange area represents the targeted mild condition range (70–120 kPa) for acetylene uses. The functionalization of **MOF-508** with $-NO_2$ or $-NH_2$ leads to a shift in the gate pressure to **c)** lower pressure or **d)** higher pressure range, respectively.

Results

Synthesis and characterization

A series of **MOF-508** derivatives composed of a mixture of **bdc** and a substituted **bdc-X** (with **X** = $-\text{NO}_2$ or $-\text{NH}_2$) ligand were synthesized with a precise control of the **bdc/bdc-X** ratio (**Figs. 2a–c** and Supplementary **Fig. 1**).¹⁷ The final linker ratio of these $[\text{Zn}_2(\text{bdc})_{2-a}(\text{bdc-X})_a(\text{bpy})]$ materials (with **a** varying from 0 to 2) was confirmed by ^1H NMR and showed almost no deviation from the linker ratio expected from the introduced mixture (**Fig. 2d** and Supplementary **Figs. 2–14**). For clarity, the samples are designated according to the function moiety (**X** = $-\text{NH}_2$, $-\text{NO}_2$) and molar percent ($n = 0\%$ to 100%) as follows, **(X)_n-MOF-508**. All materials are thermally stable up to 350°C , as observed by thermogravimetric analysis (TGA) (Supplementary **Figs. 15–17**). Next, the SEM images of as-synthesized samples show good homogeneity of crystallite sizes, which are in the micrometer range (Supplementary **Figs. 18–19**). Powder X-ray patterns, infrared analysis, and crystal structures of as-synthesized and activated samples confirmed the purity of all materials as well as their isostructural structures (Supplementary **Figs. 20–26** and **Tables 1–2**).

Interpenetrated **MOF-508** shows a dynamic transition from close (when the structure is completely evacuated from guests) to open form (when a specific gas pressure or temperature is applied).¹⁸ Thus, gas sorption properties of **(X)_n-MOF-508** series were investigated.

First, $(X)_n$ -MOF-508 samples were confirmed to behave as a solid solution, i.e., that the ligands are homogeneously distributed within the backbone structures (Supplementary Fig. 27).^{19,20} The influence of the ratio of linker mixture on the gate-opening (P_{go})/closing (P_{gc}) pressures was determined from single gas adsorption measurements performed at different temperatures and pressures (Fig. 3 and Supplementary Figs. 28–35). Acetylene physisorption of pristine MOF-508 at 273 K shows $P_{go} = 44$ kPa and $P_{gc} = 19$ kPa with an excess amount of $2.87 \text{ mol}\cdot\text{mol}^{-1}$ of acetylene adsorbed (Fig. 3).¹⁸ When $(\text{NH}_2)_n$ -MOF-508 is used, a shift in P_{go} from 48 kPa to 83 kPa was observed when n increased from 5% to 60% (Fig. 3a). In contrast, for $(\text{NO}_2)_n$ -MOF-508, the gate-opening pressure shifted to a lower pressure when n increased from 5% to 100% (from $P_{go} = 31$ kPa to $P_{go} = 1$ kPa) (Fig. 3b). The plot of the gate-opening and closing pressures as a function of molar percentage of amino groups in $(\text{NH}_2)_n$ -MOF-508 suggests that P_{go} increases exponentially

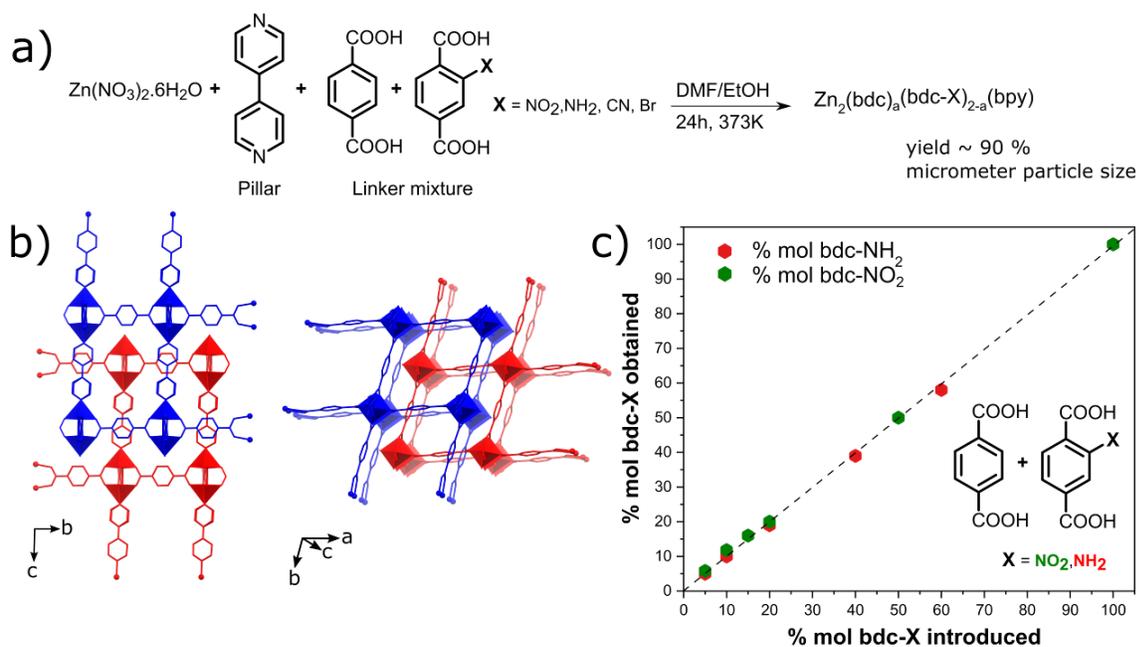


Fig. 2 | a) Synthetic scheme of $(X)_n$ -MOF-508 materials. b) (Left) crystal structure of activated MOF-508 along the b axis and (right) crystal structure of activated MOF-508 along the c axis showing a pillar ligand connecting the 2D layers. c) Molar percentage of the **bdc-X** linker within the material obtained as a function of the molar percentage of **bdc-X** introduced in the reaction. The obtained ratio was estimated from the ¹HNMR spectra of digested sample.

with the amount of **bdc-NH₂** incorporated (**Fig. 3c**). At a higher molar percentage ($n > 60\%$) of **bdc-NH₂**, P_{go} is not observed below 100 kPa; however, the value can be extrapolated from the exponential fit of the lower molar percentage data points and later confirmed by the high pressure measurements (**Fig. 3 and Supplementary Figs. 38–42**). This extrapolation is possible because the same exponential trend is observed at a lower temperature (195 K) where all P_{go} values can be determined even for high amino loading (Supplementary **Figs. 28–29**). Additionally, the introduction of nitro moiety shows a more substantial impact on the shift of the gate-opening pressure with a more pronounced exponential decay of P_{go} (**Fig. 3d**).

To rationalize the nature of the impact of linker substitution on the thermodynamics of the gate-opening process, we applied theoretical thermodynamic models for the adsorption of guest molecules in flexible nanoporous materials to the adsorption isotherms of MOF-508 and its derivatives.^{21,22} Researchers previously showed that a simple two-phase model provides a good description of gate opening in interpenetrated frameworks, where gate opening is an adsorption-driven first-order phase transition from a nonporous phase to a microporous phase. Here, we described the “open phase” adsorption by Langmuir-type fits of isotherms for all materials at 273 K. Based on the adsorption and desorption isotherms, we can calculate the value of free energy difference ΔF associated with the gate-opening process (see Supplementary Information for the equations).

First, we show that **MOF-508** exhibits a relatively small free energy difference ($\Delta F = 5.2$ kJ/mol) between the two phases (here, all energies are reported per unit cell). This value is slightly larger, but of a similar order of magnitude, than those of other gate-opening materials (e.g., [Cu(4,4'-bpy)(dhbc)₂]H₂O has $\Delta F \sim 4$ kJ/mol).^{21,23} Next, the relatively small free energy difference between the two phases explains why the material readily opens upon adsorption of small guests.

Secondly, we see that both the introduction of $-\text{NO}_2$ and $-\text{NH}_2$ substituents leads to a decrease in ΔF (see Supplementary **Table 3**). The impact of $-\text{NH}_2$ is determined to be smaller than that of $-\text{NO}_2$. The presence of nitro groups decreases the free energy difference between the closed and open phases because of the steric hindrance in the closed phase. Owing to a smaller energetic cost for gate opening, opening is easier and occurs at lower gas pressure. This is particularly noticeable in the $(\text{NO}_2)_{100}\text{-MOF-508}$ phase, where the free energy difference is almost zero, which means that the material is truly bistable, and the gate-opening transition is triggered by minute amounts of adsorbate.

In the presence of $-\text{NH}_2$ groups, the observed effect is more complex. The introduction of $-\text{NH}_2$ groups in **MOF-508** decreases ΔF , but it does not lead to a decrease in gate opening (and gate closing) pressure. This effect is smaller than for $-\text{NO}_2$ groups because $-\text{NH}_2$ is a less bulky substituent. Therefore, change in ΔF is not the dominant factor, and the driving force in this case is the evolution of affinity in the open phase, measured by the Henry constant K , which is related to the adsorption enthalpy ΔH as $K \propto \exp(-\Delta H / RT)$. When $-\text{NH}_2$ groups are introduced, the surface area of the open phase is reduced, and with it, the affinity for the guest is also reduced (K decreases). In turn, lower host–guest affinity (weaker interaction) means that it takes a higher pressure for the gas to effectively adsorb, and therefore to trigger gate opening. This explains the counterintuitive experimental observations, i.e., that the introduction of $-\text{NH}_2$ groups leads to an increase in gating pressure, while $-\text{NO}_2$ groups lead to a decrease in gating pressure. Finally, thermodynamic models also explain why the evolution of gate-opening pressure exhibits an exponential dependency on the composition of MOF ligands. In a gate-opening material, the transition pressure is determined at thermodynamic equilibrium by the following:

$$P_{go} = \frac{N}{K} \left(e^{-\frac{\Delta F}{NRT}} - 1 \right) = \frac{N}{K_0} e^{\frac{\Delta H}{RT}} \left(e^{-\frac{\Delta F}{NRT}} - 1 \right)$$

where N is the saturation uptake in the open phase; K is the Henry constant; ΔH is the adsorption enthalpy, and ΔF is the free energy difference between the phases. While the general expression

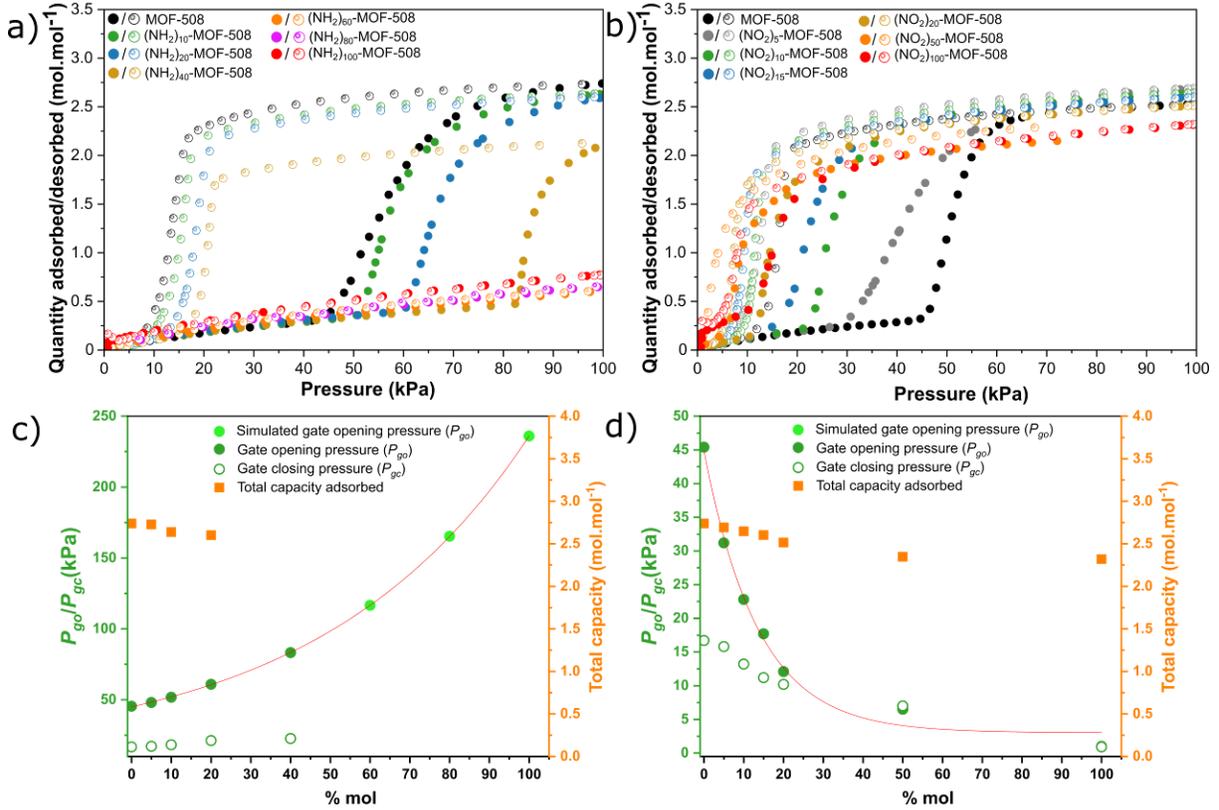


Fig. 3 | a) Acetylene (C_2H_2) adsorption isotherms for all $(NH_2)_n$ -MOF-508 at 273 K. The MOF-508 isotherm (black) was added for comparison. b) Acetylene (C_2H_2) adsorption isotherms for all $(NO_2)_n$ -MOF-508 at 273 K. The MOF-508 isotherm (black) was added for comparison. c) Acetylene (C_2H_2) gate opening (green filled circle), predicted gate opening (light green filled circle) and closing (green empty circle) pressure for all $(NH_2)_n$ -MOF-508 at 273 K. Total capacity (orange filled square) as a function of mol% of $-NH_2$ in MOF-508 if reached below 100 kPa. d) Acetylene (C_2H_2) gate opening (green filled circle) and closing (green, empty circle) pressure for all $(NO_2)_n$ -MOF-508 at 273 K. Total capacity (orange filled square) as a function of mol% of $-NO_2$ in MOF-508 if reached below 100 kPa.

is complex, in the case of the MOF-508 materials, the variation in N is marginal, and the gate-

opening pressure is controlled by the exponential dependence on both host property (ΔF) and host–guest interaction (ΔH); both quantities vary linearly with the linker composition.

These results suggest the possibility to rationally control the gate-opening and closing pressure by introducing a different ratio of functional groups. Additionally, the dependence of gate-opening and closing on the observed molar ratio of amino or nitro functional groups is not unique to acetylene but also occurs for CO₂, O₂, and CH₄, which demonstrates the versatility of the method (Supplementary **Figs. 30–35**).

Of note, the total adsorbed capacity does not considerably change for all (NH₂)_n-MOF-508 compared to that for pristine MOF-508 for acetylene gas at 273 K (**Fig. 3c**). The total adsorbed capacity at higher loading in the (NH₂)_n-MOF-508 series was recorded at a lower temperature (T = 195 K), and it demonstrated that the total capacity did not decrease for all (NH₂)_n-MOF-508 series (Supplementary **Fig. 28–29**). Conversely, the capacity decreases up to 15% for (NO₂)₁₀₀-MOF-508 (from 2.73 mol.mol⁻¹ to 2.32 mol.mol⁻¹). This difference is expected owing to the bulkier nitro moiety (VDW surface area **bdc-NO₂** = 249.2 Å²) compared to the amino one (VDW surface area **bdc-NH₂** = 223.3 Å²).

***In-situ* PXRD**

To further investigate the effect of substituted-**bdc**, the phase transformation of the structure was also characterized for MOF-508, (NH₂)₁₀₀-MOF-508, and (NO₂)₁₀₀-MOF-508 by *in-situ* PXRD under various acetylene pressures at 195 K (Supplementary **Fig. 36**).

Increasing gas pressure leads to the transition from close to open phase structure, which occurs at different pressures for the three samples. The gate-opening pressures, which correspond to the phase transition observed on the PXRD patterns, are observed at P_{go} = 0.04, 0.21, and 1.16 kPa for

(NO₂)₁₀₀-MOF-508, MOF-508, and (NH₂)₁₀₀-MOF-508, respectively, and this difference is explained by the difference in the starting structure. At a low pressure (below P_{go}), the PXRD patterns of activated MOFs (closed form) for (NH₂)₁₀₀-MOF-508 and (NO₂)₁₀₀-MOF-508 show a decrease and increase of the volume cell, respectively, compared to pristine MOF-508, as seen from the shift of the 100 peak. These results are confirmed by the crystal structure obtained from X-ray diffraction experiments (Supplementary **Table 1–2**).²⁴ The cell volume of activated (NO₂)₁₀₀-MOF-508 (715.9 Å³) is higher compared to those of MOF-508b (703.6 Å³) and (NH₂)₁₀₀-MOF-508 (701.9 Å³), which is attributed to i) the softer interaction of –NO₂ groups with the metal nodes and ii) the steric hindrance of –NO₂ moiety, which further separates the aromatic part of the ligand. Therefore, the stabilization of the structure through π – π interactions is reduced, as observed in pristine MOF, which explains the lower observed gate-opening pressure and is supported by the decrease of the free energy when increasing the nitro content, as described above.

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For (NH₂)₁₀₀-MOF-508, the higher pressure needed for the gate opening to occur is explained from the initially more contracted structure owing to interactions between –NH₂ moieties and oxygen atom from the metal paddlewheel.²⁴ Finally, in the higher pressure range, all MOF structures are completely open, and they display similar diffraction patterns (Supplementary **Figs. 36d–f**). Considering that the opened and closed forms of each materials are isostructural, the difference in the gate-opening pressure is attributed to the functionalized **bdc-X** linker.

Optimal conditions

Owing to the insight on the structural dynamics of MOFs obtained from the combination of various characterization techniques, we can predict the behavior of (X)_n-MOF-508 materials and determine optimal conditions for acetylene adsorption/desorption.

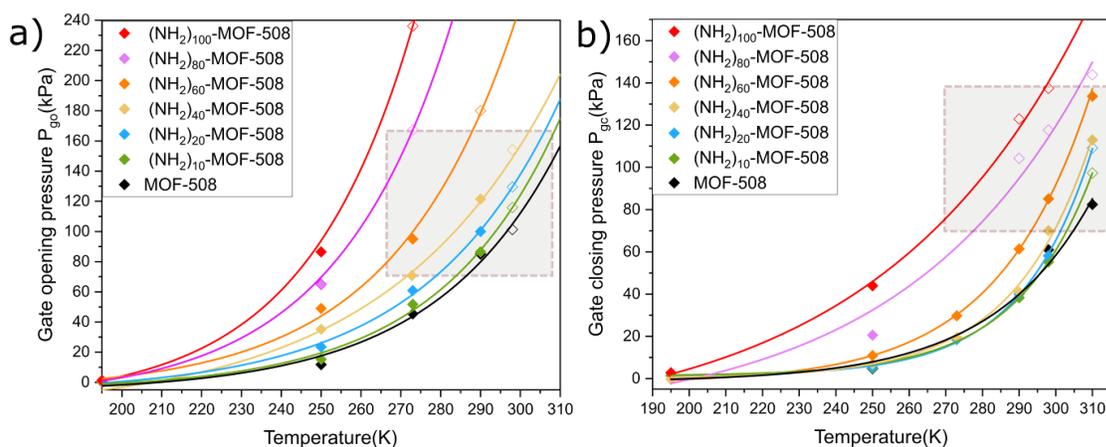


Fig. 4 | Predicted gate a) opening and b) closing pressures of **MOF-508** and **(NH₂)_n-MOF-508** as a function of temperature. Gray square shows the best region for practical temperature and pressure use conditions for acetylene gas storage (temperature of 273–310 K and pressure of 70–160 kPa).

The collected measurements combined with estimated data allow us to predict gate opening and closing as a function of temperature (**Fig. 4**). Thus, engineering our system for the optimal temperature and pressure required for safe handling and use of acetylene gas is possible ($T = 273\text{--}310\text{ K}$ and $P = 70\text{--}160\text{ kPa}$, gray area, **Fig. 4**). In terms of the ambient condition utilization of the bottle, **(NH₂)₆₀-MOF-508** shows interesting results with $P_{go} = 95\text{ kPa}$ at 273 K and $P_{gc} = 91\text{ kPa}$ at 298 K, which meets the practical use for safe acetylene storage (**Fig. 4 and Supplementary Figs. 38–43**). Furthermore, when comparing the usable volumetric capacity (UVC) of acetylene for **(NH₂)₆₀-MOF-508** with those of reported MOFs, the former provides an UVC of 106 v/v, with complete filling at 100 kPa and 273 K and 84% degassing at 70 kPa and 298 K, which is higher than for most reported MOFs (Supplementary **Table 4** and **Fig. 42**).^{26–29} Of note, compared to all reported MOFs for acetylene storage, the **MOF-508** family is the only material with gate-opening behavior and, thus, paves the way for the future development of flexible MOF for acetylene storage. In addition, recording adsorption isotherms at a higher pressure than at an ambient one was possible with a higher amino content ($x > 60\%$), to observe complete gate-opening pressure while

staying under the 2 bar limit pressure, which shows the potential for obtaining better UVC. This pressure range, which makes the industrial application of these MOFs possible, is unprecedented.

In summary, compared to the current solvent compression method, the use of MOF adsorbents to store acetylene exhibits practical advantages. First, high acetylene purity can be achieved because no solvents (acetone or DMF) that can degrade or contain contaminants are required. Second, avoiding flammable and harmful solvents increases safety when handling acetylene bottles. In addition, the adsorbed gas can be confined in MOF pores at a lower pressure compared to 19 bar at 20°C needed for the non-static solvent/porous filler storage method. Furthermore, functionalized flexible **(X)_n-MOF-508** can compress large quantities of acetylene (~ 129 v/v) in less than 2 bar, which allows to develop lighter gas bottle containers. Finally, the remaining unused acetylene, which requires low pressure (< 70 kPa) to be completely desorbed, presents a potential explosive risk, which can be solved by flexible MOFs. Indeed, we demonstrated the rational design of MOF whose gate-opening pressure is controlled by varying the ratio of ligand substituents. Therefore, flexible **(X)_n-MOF-508** offers tunable uptake/release gas pressures, while being also able to fully release acetylene at an ambient pressure (storage at 273 K and P > 2 bar, release at 298 K and P ≤ 70 kPa). This approach to gas storage demonstrates its practicality through the design of a narrow pressure range of gas introduction and release and with a low pressure limit for pressure-sensitive unstable gas substance.

These results open the way for the application of flexible MOF to the effective storage of targeted gases from the discovery of cooperative adsorption phenomena.

Methods

MOF synthesis

MOF-508 and **(X)_n-MOF-508** series ($n = 0\%$ to 100% and $\mathbf{X} = -\text{NH}_2, -\text{NO}_2$) were prepared following the previously used method.¹⁷ 1,4-benzenedicarboxylic acid is dissolved with functionalized 1,4-benzenedicarboxylic acid (total 2 eq) in DMF (concentration [**bdc linker**] = 0.15 M) at 100°C for 15 min. An ambient solution of Zn(NO₃).6H₂O (2 eq) in DMF (concentration [**metal**] = 0.4 M) was added to the stirring mixture at 100°C for 10 min. 4,4'-bipyridine (1 eq) is dissolved at room temperature in EtOH (concentration [**pillar**] = 0.1 M) and added to the reaction mixture. The mixture was kept at 100°C for 24 h. Next, after cooling down to room temperature, the resulting precipitate was collected by centrifugation and washed three times with DMF (3 × 40 mL) and three times with EtOH (3 × 40 mL). The white/yellow powder is dried under vacuum at 50°C for 12 h.

Gas sorption

Single gas adsorption isotherms of MOF samples were previously activated through thermal activation at 423 K for 12 h before measurement. The adsorption isotherms were measured with BELSORP-CRYO (C₂H₂ and CO₂) and BELSORP-MAX (C₂H₂, CO₂, CH₄, and O₂) volumetric adsorption equipment from Bel Japan. Targeted relative pressures in the range of 0.01–100 kPa were defined, and limits of excess and allowance amount were set to 5 and 10 cm³ g⁻¹, respectively. The equilibration condition for each point was 0.03% pressure change within 300 s. Finally, the dead volume was determined using helium gas.

In-situ PXRD measurements

Gas sorption measurements were performed on BELSORP-18PLUS (MicrotracBEL, Japan, Corp.) automated volumetric sorption analyzers equipped with cryostat temperature controllers. The *in-situ* PXRD/adsorption measurements were performed using a Rigaku SmartLab (with Cu-K α radiation) connected to BELSORP-18PLUS volumetric adsorption equipment. These instruments were synchronized with each other, and each PXRD pattern was obtained at each point of sorption isotherms. Finally, the samples were previously activated through external thermal activation at 423 K for 12 h and further 2 h internal activation at 423 K before measurement.

Acknowledgements

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Author Contributions

M.B. S.K., C.L and R.R. formulated the project. M.B and C.L. synthesized the compounds and collected the gas adsorption data. M.B. collected and analyzed all the ¹H NMR data. M.B. collected all the *in-situ* PXRD data. M.B. and C.L analyzed all adsorption data. M.B and C.L. collected and analyzed the TGA and powder X-ray diffraction data. M.B. and K.S. collected and analyzed the single-crystal X-ray diffraction data. A.L. and M.B. collected all SEM images UV-vis and IR data.

F-X.C. performed the thermodynamics calculations. T.O. and C.L. build and collected the sorption data from the acetylene experimental set-up. M.B with assistance from O.K-I, A.L. and F-X.C wrote the paper, and all authors contributed to revising the paper.

Data availability

Reprints and permissions information is available at www.nature.com/reprints. The authors declare competing financial interests: details are available in the online version of the paper. Crystallographic data for as-synthesized and activated (**NO₂**)₁₀₀-**MOF-508** structures have been deposited in the Cambridge Structural Database with CCDC numbers 2036574 - 2036575.respectively. Readers are welcome to comment on the online version of the paper. Correspondence and requests for materials should be addressed to S.K.

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Figure legends

Fig. 1 | Comparison of acetylene gas sorption profiles and usable capacity in the pressure range of 0–100 kPa for **a**) a rigid material showing a type I Langmuir isotherm, **b**) flexible interdigitated **MOF-508** showing a S-shape isotherm with a gate-opening pressure (P_{go}) and a gate closing pressure (P_{gc}). The orange area represents the targeted mild condition range (70–120 kPa) for acetylene uses. The functionalization of **MOF-508** with –NO₂ or –NH₂ leads to a shift in the gate pressure to **c**) lower pressure or **d**) higher pressure range, respectively.

Fig. 2 | **a**) Synthetic scheme of (X)_n-**MOF-508** materials. **b**) (Left) crystal structure of activated **MOF-508** along the b axis and (right) crystal structure of activated **MOF-508** along the c axis showing a pillar ligand connecting the 2D layers. **c**) Molar percentage of the **bdc-X** linker within the material obtained as a function of the molar percentage of **bdc-X** introduced in the reaction. The obtained ratio was estimated from the ¹HNMR spectra of digested sample.

Fig. 3 | **a**) Acetylene (C₂H₂) adsorption isotherms for all (NH₂)_n-**MOF-508** at 273 K. The **MOF-508** isotherm (black) was added for comparison. **b**) Acetylene (C₂H₂) adsorption isotherms for all (NO₂)_n-**MOF-508** at 273 K. The **MOF-508** isotherm (black) was added for comparison. **c**) Acetylene (C₂H₂) gate opening (green filled circle), predicted gate opening (light green filled circle) and closing (green empty circle) pressure for all (NH₂)_n-**MOF-508** at 273 K. Total capacity (orange filled square) as a function of mol% of –NH₂ in **MOF-508** if reached below 100 kPa. **d**) Acetylene (C₂H₂) gate opening (green filled circle) and closing (green, empty circle) pressure for all (NO₂)_n-**MOF-508** at 273 K. Total capacity (orange filled square) as a function of mol% of –NO₂ in **MOF-508** if reached below 100 kPa.

Fig. 4 | Predicted gate **a**) opening and **b**) closing pressures of **MOF-508** and (NH₂)_n-**MOF-508** as a function of temperature. Gray square shows the best region for practical temperature and pressure use conditions for acetylene gas storage (temperature of 273–310 K and pressure of 70–160 kPa).